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Gold Nanostar-Coated Polystyrene Beads as Multifunctional Nanoprobes for SERS Bioimaging

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ABSTRACT

Hybrid colloidal nanocomposites comprising polystyrene beads and plasmonic gold nanostars are reported as multifunctional optical nanoprobes. Such self-assembled structures are excellent Raman enhancers for bio-applications as they feature plasmon modes in the near infrared "first biological transparency window". In this proof of concept study, we used 4 mercaptobenzoic acid as a Raman-active molecule to optimize the density of gold nanostars on polystyrene beads, improving SERS performance and thereby allowing *in vitro* cell culture imaging. Interestingly, intermediate gold nanostar loadings were found to yield higher SERS response, which was confirmed by electromagnetic modeling. These engineered hybrid nanostructures notably improve the possibilities of using gold nanostars as SERS tags. Additionally, when fluorescently labeled polystyrene bead are used as colloidal carriers, the composite particles can be applied as promising tools for multimodal bioimaging.

INTRODUCTION

Surface-enhanced Raman scattering (SERS) is a powerful spectroscopic technique that has attracted large interest due to numerous advantages over conventional analytical tools.¹ It provides non destructive characterization, high sensitivity even down to single molecule detection, $2,3$ and multiplexed capability based on narrow and sharp molecule-specific vibrational Raman peaks. ⁴ SERS relies on the unique optical properties of metallic nanostructures, which can support high electromagnetic fields upon excitation of localized surface plasmon resonances (LSPRs), in turn enhancing the Raman scattering of adsorbed molecules. Recent work has focused on the development of novel plasmonic nanostructures for SERS (usually known as SERS substrates). In particular, gold nanoparticles (Au NP) are extensively used due to versatile synthesis methods, long-term stability and biocompatibility.⁵ The ideal SERS substrate should be reproducible, chemically stable and exhibit high and spatially uniform SERS efficiency. Extremely high electromagnetic (EM) field enhancements (also named "hot spots") are produced at the junctions between adjacent metal nanoparticles, 6 as well as at sharp edges in anisotropic nanostructures.⁷⁻⁹ Among the latter, gold nanostars (Au NSs) are of especial interest since their optical responses can be readily tuned within the visible and near infrared (NIR) regions and their spikes act as efficient nanoantennas so that high EM fields are generated at the tips, giving rise to multiple intrinsic hot spots within a single nanoparticle.^{10,11} Various Au NS-based SERS substrates have been reported for chemical detection. Extremely high Raman signals have been obtained for dithiolated Raman-active molecules trapped between Au tips and a smooth gold film, 9 which were explained due to enhanced confinement effects.¹² An interesting alternative for practical applications comprised the immobilization of Au NSs on either gold or flexible polydimethylsiloxane substrates for chemical sensing of analytes that can directly interact with the Au NS surface.^{13,14} Alternatively, colloidal templates have been also widely

employed as plasmonic supports. Polystyrene (PS)-based plasmonic SERS substrates can be easily manipulated, are resistant to particle aggregation and may display high SERS activity.¹⁵ PS is one of the most commonly used polymeric materials due to its low toxicity, high water stability and easy synthesis and surface functionalization.¹⁶ Although *in situ* growth of nanoparticles onto the surface of PS microspheres has been widely reported, $17-20$ significantly better control over the desired NP size and shape can be achieved when Au NPs are pre-synthesized and subsequently assembled onto the polymer surface. 2^{1-23}

Herein we report the assembly of Au NSs onto PS beads with tunable Au NS density, which can be used to optimize the SERS efficiency of these hybrid nanomaterials. We additionally employed highly detailed electromagnetic modeling as a complementary tool to analyze and select the most efficient configuration of these hybrid nanostructures toward applications in cell bioimaging. In this context, we propose the combination with fluorescence imaging. The integration of two functionalities in the same composite particle is proposed to compensate the main deficiencies of both techniques and constitutes a major strategy to independently track and image objects using different optical tools.²⁴ Although fluorescence is a well-established technique that allows fast imaging acquisition, 2^5 fluorophores can show spectral overlap, photo-bleaching and are typically limited to visible light excitation.²⁶ On the other hand, SERS offers the possibility of extended multiplexing and using excitation wavelengths at the so-called "biological transparency window".²⁷ In this respect, fluorescent-SERS tags have successfully used for imaging of duplex co-expressed markers on cancer cells,²⁸ as well as for simultaneous multimodal tumor detection and photodynamic therapy *in vivo*. ²⁹ Although most of the reported strategies comprise the conjugation of a fluorophore onto the plasmonic core, $30-32$ we alternatively propose the use of fluorescently-labeled PS beads as carriers to attach the SERS-active Au NSs. Compared to previously reported fluorescent-SERS dual mode tags, our multifunctional nanoprobes are more robust due to the use of PS supports¹⁵ and the co-localization of the plasmonic nanostars on the colloidal structure allows the detection of the intracellular SERS signal. Additionally, NIR excitation can be used and silica coating is avoided, so that the stability in biological media can be notably improved.^{33,34} We demonstrate that our designed nanoassemblies are promising nanotools for multimodal bioimaging combining three different spectroscopic methods, namely fluorescence, light scattering and SERS.

MATERIALS AND METHODS

Materials

Milli-Q water (resistivity 18.2 MΩ·cm) was used in all experiments. Hydrogen tetrachloroaurate trihydrate (HAuCl4·3H2O, ≥99.9%), sodium citrate tribasic dihydrate (≥98%), silver nitrate (AgNO3, ≥99%), L-ascorbic acid (AA, ≥99%), O-[2-(3 mercaptopropionylamino)ethyl]-O'methylpolyethylene glycol (PEG-SH, Mw 5,000 g/mol), 4-mercaptobenzoic acid (4-MBA, 90%), were purchased from Sigma-Aldrich. Hydrochloric acid solution (HCl, 37%) was purchased from Panreac. Fluorescently labeled aminofunctionalized PS beads ($\lambda_{ex}/\lambda_{em}$ = 477nm/519nm; NP size 483 \pm 3 nm) were purchased from IKERLAT Polymers (Lasarte, Spain). Non-labeled carboxylic acid-functionalized PS beads were a gift from the same company. All glassware was washed with aqua regia, rinsed 3-fold with milli-Q water and dried before use. Dulbecco´s Modified Eagle´s Medium (DMEM), fetal bovine serum (FBS), penicillin-streptomycin (PStrep), trypsin-EDTA, wheat germ agglutinin-AF647 (WGA647) and 4,6-diamidine-2-phenylindole dihydrochloride (DAPI) were purchased from Invitrogen. MTT cell proliferation kit was purchased from Roche.

Sample Characterization

Transmission electron microscopy (TEM) images were collected with a JEOL JEM-1400PLUS TEM operating at 120 kV, using carbon coated 400 square mesh copper grids. HAADF-STEM images and electron tomography tilt series were acquired using a double aberration corrected cubed FEI Titan 50-80 electron microscope operated at 300 kV. For the reconstruction of the series the SIRT algorithm was used, as implemented in the ASTRA toolbox.³⁵ Vis-NIR optical extinction spectra were recorded using an Agilent 8453 UV-Vis diode-array spectrophotometer.

Synthesis of Gold Nanostars

Au NSs were prepared by a modified seed-mediated growth method.³⁶ Briefly, the seed solution was prepared by adding 5 mL of a 1 % citrate solution to 95 mL of boiling 0.5 mM HAuCl4 solution under vigorous stirring. After 15 min of boiling, the solution was cooled down to room temperature and then kept at 4 °C for long-term storage. The as-synthesized Au nanoparticle seeds had an LSPR maximum at 519 nm. For Au NS synthesis, 500 μL of the citrate-stabilized seed solution was added to 10 mL of HAuCl4 (0.25 mM) solution containing 10 μL of HCl (1.0 M) in a 20 mL glass vial at room temperature, under moderate stirring. Quickly, 100 μL of AgNO₃ (3 mM) and 50 μL of AA (100 mM) were added simultaneously to the above solution. The solution rapidly turned from light red to greenish indicating the formation of Au NSs. Immediately after synthesis, the solution was stirred with PEG-SH (final concentration $8 \cdot 10^{-7}$ M) for 15 min, washed by centrifugation (1190 g, 25 min, 10° C) and redispersed in water. The Au^0 concentration of the Au NS solution was adjusted to 0.5 mM.

Assembly of Au NSs and PS beads

A simple procedure was used to form Au NS-coated PS particles. 1mL of PEG-stabilized Au NS solution 0.5 mM was added dropwise onto 100 µL of PS in an Eppendorf tube under sonication and the particle mixture was incubated overnight. Three different initial PS

concentrations (0.15, 0.3 and 1 % in solids) were used to obtain *high*, *medium* and *low* Au NS loading, respectively. The unbound Au NSs were removed by two centrifugation steps (1770g, 3 min, 20 °C) and the samples were redispersed in Milli-Q water adjusting the concentrations to $1.5 \cdot 10^{-2}$ (PS, % in solids).

Cell Culture

Human lung epithelial cancer (A549), breast cancer (MCF-7) and murine monocytemacrophage (J774) cell lines were grown in DMEM supplemented with 10% FBS and 1% PStrep (known as cDMEM). Cells were passaged using Trypsin-EDTA. For SERS, confocal and dark field microscopy, between $5 \cdot 10^4$ to $2 \cdot 10^5$ cells were plated in glass bottomed dishes with marked grids ("Grid-50", Ibidi) and allowed to adhere overnight. Fluorescently labeled Au NS-PS particles were added to cells at a final concentration of PS from 7.5·10⁻⁴ to $3.75 \cdot 10^{-4}$ % in solids, depending on the experiment setup, and left for various time points at 37 °C to allow particle uptake.

Confocal Fluorescence and Dark-Field Microscopy

Particle uptake in A549 cells was imaged using a Zeiss LSM confocal microscope with 477 nm excitation and a 500-550 nm BP filter. Either 40x (NA=1.3) or 63x (NA=1.4) objectives were used. Multichannel (transmission and fluorescence) Z-stacks were taken using ZEN software. For dark field microscopy (DFM) images, cells were fixed using 4 % formaldehyde (20 min, room temperature) and mounted using a #1.5 coverslip and mounting fluid (Dako). Once dry, the dish was inverted and viewed using a Nikon Eclipse Ti-U microscope equipped with a 60x (NA=0.7) objective and oil condenser. Neutral density and neutral color balance filters were inserted. Photos were taken using a 0.1 s exposure and light intensity levels and other variables were unchanged for the different samples.

Cell Viability

Nanoparticle induced cytotoxicity was measured using the MTT assay. A549 cells were plated in 96 well plates at $1 \cdot 10^4$ cells/well and allowed to adhere before adding NPs at final PS concentrations ranging from $1.5 \cdot 10^{-3}$ to $9.4 \cdot 10^{-5}$ % in solids. Cells were incubated for 66 h before removal of the cell medium and addition of MTT reagent (1/20 dilution in cDMEM). Cells were incubated for 1.5 h at 37 ºC, followed by removal of the supernatant and addition of DMSO to lyse cells. Absorbance was read at 550 nm and cell viability calculated as a % of the control.

SERS Measurements

SERS was measured with a confocal Raman microscope (Renishaw inVia) equipped with 1024×512 CCD detector using a 785 nm laser excitation source (maximum output of 250 mW) and a 1200 l/mm diffraction grating. For reference and test SERS experiments, 1 mL of 4-MBA labeled Au NS-PS beads (*low*, *medium* and *high* Au NS loading, final 4-MBA concentration 1 mM) was placed in a 1 mL glass vial (Thermo Fisher) and spectra were collected using a $10\times$ objective (NA=0.35) in expanded scan mode with an integration time of 10 s and a laser power of 18 mW. SERS measurements in cells were carried out in static mode (center of scattered wavenumber 1200 cm^{-1}) with an integration time of 1s and a laser power of 15 mW using a water immersion 40x objective (NA=0.85). Maps were generated by plotting the intensity of the 4-MBA ring deformation (1078 cm^{-1}) as a function of grid position. For estimation of the SERS enhancement factor, 4-MBA labeled Au NS-PS beads (*low*, *medium* and *high*, 50 µL) were dropped onto glass slides and a thin film was formed by spin coating (30 s, 33 rps). Solid 4-MBA was pestled and compacted onto a glass slide to form a homogeneous, thick layer. SERS and Raman spectra were measured with integration time of 10 s (extended mode) at laser power of 1.7 mW using a 50x objective (NA=0.75). The average laser powers were measured in air after passing the laser beam through the objectives used in the corresponding experiments by means of a photodiode sensor (PD300- 3W) and a power meter (Nova, both from Ophir).

Electromagnetic-Field and SERS Intensity Calculations

Spectral and near field enhancement simulations were carried out using a numerical method based on a surface integral equation-method of moments (SIE-MoM) formulation $(M³$ solver).^{37,38} With this methodology, the metallic nanostructures are replaced by equivalent electric and magnetic currents placed over the particle boundary surfaces and interfaces, thus reducing the computation domain of any radiation or scattering problem to the surface boundaries of materials. Additionally, the SIE-MoM approach yields high stability, particularly when dealing with resonant metallic response, as the singular behavior of fields is analytically described by the Green's function and its derivatives. Through Maxwell's equations and the boundary conditions for the total fields we derive a set of SIEs for the unknown equivalent currents. These SIEs are subsequently discretized by applying a Galerkin MoM procedure in terms of a set of basis and testing functions, leading to a dense $N \times N$ matrix system of linear equations (being *N* the number of basis functions used to expand the unknown equivalent currents on the surface boundaries).

For a realistic simulation of the large-scale plasmonic systems considered here, SIE-MoM was accelerated via the multilevel fast multipole algorithm-fast Fourier transform (MLFMA-FFT),39,40 providing a high algorithmic efficiency-computational cost of *O*(*N* log *N*)—along with high-scalability via parallelization using multicore computer clusters.

The simulations were carried out on a workstation with four 15-core Intel Xeon E7- 4880V2 37.5 MB L3 cache processors at 2.5 GHz. A high surface mesh density was considered using up to 975,840 basis functions for the most computationally intensive case (Au NS-PS particles with high loading), with an edge size ranging from 1/162 wavelengths for the shortest simulated wavelength to $1/405$ for the longest. A relative error norm of 10^{-5}

was considered to halt the Krylov iterative solver. Considering the above, the computation times ranged from 21 minutes to 1.5 hours per wavelength, while the memory use ranged from 11 to 28 GB.

RESULTS AND DISCUSSION

Monodispersed PS beads coated with Au NSs were prepared using a two-step procedure: PEG-coated Au NSs were first synthesized through a well known seed-mediated growth method³⁶ and subsequently allowed to self-assemble onto the amine-terminated PS surface¹⁴ (**Figure 1**). This strategy leads to highly stable nanostructures and allows tight control over the final plasmonic performance. An alternative method, based on the adsorption of Au nanospheres on PS and *in situ* growth of gold tips, resulted in aggregation and growth of nonuniform particles containing branched structures of different sizes and shapes that protrude out from the polymer surface (**Figure S1a, SI**). It should also be noted that, when PS spheres functionalized with carboxylic acid were used, Au NSs did not bind to the polymer surface (**Figure S1c, SI**). 22

Figure 1 Upper panel: Schematic illustration of gold nanostar synthesis (**i**) and assembly (**ii**). Lower panel: representative TEM image of a Au NS-PS bead (**a**) and UV-Vis-NIR spectra (**b**) of Au NS (black line) and Au NS-PS (red line) colloids.

Due to the strong affinity of amino groups with the gold surface, the polymer beads were densely covered with Au NSs, as confirmed by TEM (**Figure 1a**). In addition, annular dark field - scanning TEM (ADF-STEM) analysis was carried out for detailed analysis of individual self-assembled particles. As can be observed in **Figure 2a** a central polymer particle is surrounded by smaller and brighter objects that correspond to Au NSs. Electron tomography and three-dimensional (3D) reconstruction of the same particle provided additional information regarding Au NS distribution. The 3D reconstruction image (**Figure 2b**) shows that Au NSs (golden) are indeed adsorbed around the whole PS surface (blue) but the distribution is not completely uniform and a certain degree of aggregation is consistently observed. The full reconstruction movie is provided as supporting information.

Figure 2 ADF-STEM (**a**) and 3D reconstruction (**b**) images of an Au NS-PS bead.

The same method could be applied to other metal nanoparticles, such as PVP-stabilized Au $NSs¹⁰$ or silver nanoparticles $(Ag NPs)⁴¹$, as exemplified in **Figure S2 (SI)**. It is important to note that the density of Au NSs per PS bead could be controlled by simply varying the concentration of PS while keeping constant the Au NS concentration. Thus, PS spheres could be coated with *low* (**Figure 3a**), *medium* (**Figure 3b**) and *high* Au NS loadings (**Figure 3c**). It should be noted however that, when increasing excessively the Au NS concentration during the assembly process, Au NS aggregation was ultimately induced and therefore uniform coatings could not be obtained (**Figure S3, SI**). The number of Au NS on the PS surface could be estimated by analyzing at least 20 TEM images, obtaining for *low*, *medium* and *high* coatings an average of ca. 20, 80 and 120 Au NSs per PS bead, respectively (**Figure S4, SI**). Again, as the density of Au NSs increased, so did the tendency to form small clusters on the PS core. The spiky morphology of the Au NSs gives rise to an LSPR band corresponding to tip plasmon modes, as well as a less intense shoulder from the core mode.⁴² A similar optical response was found for the Au NS-PS composite particles, with some differences (LSPR red-

shift and broadening) which are due to the presence of the PS core (enhanced scattering at low wavelengths and increased refractive index), but also from the close proximity between some Au NSs on the PS surface which leads to plasmon coupling effects (**Figures 1b** and **S2**).12 We also found that, when the Au NS density was increased, the intensity of the LSPR band also increased, as shown in **Figure 3d**. A similar trend was observed when calculating the extinction spectra of the assembled particles (**Figure 3e**). Trying to represent the PS-Au NSs system in its full complexity, some degree of polydispersity was considered for the assembled Au NSs in the numerical simulations, which were modeled as 12 sharp tips branching out from a central spherical core. Based on representative TEM images of the Au NSs, a (normally distributed) branch length of 22 ± 8 nm was considered, with a width of 15 nm at the base and a tip apex diameter of 4 nm. The core diameter was fixed at 35 nm. Regarding the PS sphere, the diameter was set to 483 nm. The Au NSs were randomly distributed around the PS beads, but some aggregation (slightly higher as the loading increases, see **Figure 5**) was induced, again based on TEM images. A minimum distance of 1 nm, both between neighboring Au NSs and between Au NSs and the PS bead, was considered. With the above prescriptions, ten different models were simulated and averaged to obtain the spectrum in each of the three different density cases. The calculated spectra are in close agreement with the experimental results, with LSPR red-shift and broadening, as well as enhanced scattering at low wavelengths, due to the presence of the PS core and plasmon coupling effects between Au NSs.

Figure 3 Representative TEM images (**a**-**c**) and experimental (**d**) and calculated (**f**) Vis-NIR spectra of PS beads coated with different Au NS densities (blue = *low*, red = *medium* and black = *high*).

One the other hand, large EM field confinements are expected within nm-sized gaps between neighboring Au NSs, which individually support hot spots at their tips.^{9,43} Therefore, high SERS signals are expected for Raman-active molecules trapped at such strong localized EM fields.⁴⁴ We used 4-MBA as Raman reporter molecule, which provides high SERS signal under 785 nm excitation and is known to strongly bind to the Au surface. In this case, the presence of MBA is particularly useful, as it has been not only been reported to prevent Au NS reshaping through thiol binding, $45,46$ but additionally the negative charges at the carboxylic groups oppose particle aggregation.⁴⁷ The SERS response from 4-MBAlabeled nanocomposites was measured and compared for particles with different Au NS loadings. **Figure 4a** depicts representative SERS spectra of 4-MBA from PS beads coated with *low*, *medium* and *high* Au NS densities. As expected, the most prominent peaks appear at 1580 and 1078 cm-1 corresponding to ν(C−C) ring stretch and ring breathing modes of the

molecule, respectively.48–50 Remarkably, the sample with the highest Au NS density (*high* Au NS) did not result in the most intense SERS signal. A similar behavior has been previously reported for aggregated Au NSs and was explained by some sort of hot-spot deactivation due to interaction between tips with different orientations and geometries.^{47,51} On the one hand, this situation is in stark contrast to the aggregation-induced hot-spot formation between e.g. spherical nanoparticles or nanorods.⁵² A recent publication by our group also showed that a monolayer of densely packed spherical Au NPs generates a significantly higher SERS signal than a dense monolayer of Au NSs with similar particle size.⁴⁶

The experimental enhancement factor (EF) for assemblies of different Au NS coverage can be calculated using the following equation:

$$
EF = \frac{I_{SERS} \times N_{bulk}}{N_{ads} \times I_{Raman}}
$$

where *I_{SERS}* is the intensity of a vibrational mode in the SERS spectrum, *I_{Raman}* the intensity of the same mode in the bulk Raman spectrum, N_{ads} the number of molecules adsorbed onto the nanoparticle surface contributing to the SERS signal and N_{bulk} the number of molecules sampled in the bulk material. Ideally, the Raman spectrum of 4-MBA in aqueous solution would be used to determine the EF for AuNS-PS assemblies in solution, but no vibrational bands of 4-MBA could be observed under these conditions. Thus, the Raman spectrum of 4- MBA was measured from solid powder and used to normalize SERS spectra measured from a dried monolayer of PS beads with different AuNS coverage. The number of sampled 4-MBA molecules from the bulk was estimated by the method reported by Khan et al.,⁵³ whereas the number of sampled molecules from SERS was calculated from the Au NS surface available for each coverage degree within the laser illumination area, multiplied by the bonding density of 4-MBA (\approx 0.5 nmol cm⁻²),⁵⁴ assuming a complete monolayer of adsorbed molecules. Using the intensity of the $v(CC)_{ring}$ breathing (at 1102 cm⁻¹ in bulk and at 1078 cm⁻¹ for

SERS samples) EFs can be estimated resulting in 7.3×10^7 (*low*), 7.5×10^7 (*middle*) and 1.8×10^{7} (*high*) which is within the typical range between 10^{7} - 10^{8} for other gold nanoparticles.⁵⁵ A more detailed presentation of the EF estimation is provided in the Supporting Information.

Figure 4 (**a**) SERS spectra of 4-MBA adsorbed on Au NSs-PS beads with different NS densities (blue=*low*, red= *medium*, black=*high*). Spectra were background corrected and plotted after applying an offset for more convenient data presentation. The signal marked with (*) is from the solvent of the 4-MBA stock (ethanol, 880 cm⁻¹). (**b**) SERS intensity of 4-MBA breathing mode as function of the average number of Au NSs per PS bead.

Detailed understanding of the SERS performance was achieved by means of theoretical modeling of the electric near field enhancement by the assembled nanostructures at different Au NS coverage. As the SIE-MoM formulation allows us to simulate the response of objects with arbitrarily complex morphologies, we used a model using a realistic Au NS distribution on the PS bead, as previously determined from TEM images (**Figure 5a-c**). Calculated (total) SERS intensity was averaged over p-polarized and s-polarized normal irradiation for particles containing different Au NS loadings (**Figure 5e**). For both *low* and *medium* Au NS coatings, the simulations indicate high electromagnetic field enhancement at the tips of every individual Au NS, with hardly any effects of plasmon coupling. Accordingly, the overall SERS intensity increases with the total number of Au NS tips, i.e. with the Au NS density. For *high* Au NS density a different situation is however observed, where the intensity of electromagnetic field at tips within aggregated Au NS "islands" fades out, whereas hot spots remain at the boundaries only (**Figure 5c, inset**). The clusters thus behave as bigger particles with significantly reduced SERS activity at the resonant wavelength (**Figure 5e,** black curve). Indeed, this effect was not observed for simulations considering equally *high* Au NS loading but with a more uniform distribution over the PS sphere, i.e. no clustering (**Figure 5d**). The interparticle separation indeed avoids plasmon deactivation in this case, resulting in intense hot spots at all individual tips, so that the SERS intensity again increases with the number of Au NSs on the PS support, as observed for *low* and *medium* coverages (**Figure 5e** red curve). The results of our simulations are therefore in good agreement with the experimental results and provide guidance toward engineering new Au NS-based substrates for SERS. Whereas isolated Au NSs preserve their "activated" sharp tips, we demonstrate that aggregation leads to strong plasmon coupling can lead to hot spot deactivation and therefore to lower SERS intensity. This may also be related to recent results showing that aggregation within endosomes hinders the efficiency of Au NSs toward photothermal therapy.⁵⁶

Figure 5 Calculated near field enhancement maps (λ = 785 nm) of Au NS-PS particles. (a-c) Realistic modeling of Au NS-PS particles with *low* (a), *medium* (b) and *high* (c) Au NS loadings. (**d**) PS particles with *high* Au NS density and even distribution. (**e**) Total SERS intensity as a function Au NS density, for uniform Au NS distribution (red) and for partially aggregated Au NSs on PS, resembling the experimental observations (black).

As a result of experimental and theoretical optimization, we selected the most SERSefficient configuration (4-MBA labeled Au NS-PS, *medium* Au NS density) to demonstrate the applicability of these hybrid nanostructures for imaging of living cells. After centrifugation, the particles were successfully redispersed in cDMEM and incubated for 24 hours to ensure stability in biological media, as indicated by no LSPR band alteration in the Vis-NIR spectra before and after incubation (**Figure S5**, **SI**). It is worth mentioning that neither tip reshaping nor Au NS detachment from the PS surface was observed. Successful *in vitro* SERS bio-imaging requires that the nanotags are non-toxic and effectively uptaken by living cells. We evaluated the SERS efficiency of the hybrid particles as compared to free Au NSs. An equal amount of A549 cells was incubated either with 4-MBA-labeled Au NS-PS assemblies at a concentration of 7.5 $\cdot 10^{-4}$ (% PS in solids, [Au⁰] = 0.25 mM) or the equivalent

concentration of free 4-MBA-labeled Au NSs. **Figure 6** illustrates the corresponding 4-MBA SERS spectra and SERS maps obtained from both free Au NS and composite particles, upon internalization by A549 cells after 24 h incubation. The SERS fingerprint of 4-MBA was clearly identified and no interference peaks originating from cells or buffer solution under the present experimental conditions were observed. As compared to free Au NSs, the assembled nanotags yielded remarkably intense SERS signals, which may be related to enhanced uptake of NPs clustered on a delivery platform, as previously reported.⁵³ In order to confirm the cellular uptake of both Au NS-PS and Au NS, TEM images were acquired after 12 h incubation with cells (**Figure S6**, **SI**). Both types of particles were successfully internalized by the cells, most likely by endocytosis, 57 and importantly no sign was found of Au NS that could have detached from the PS beads (**Figure S6a**, **SI**), indicating the high stability of the hybrid nanostructures. We additionally explored the general application regarding the choice of cell lines *via* SERS imaging of human MCF-7 breast cancer cells and murine J774 macrophages, with equally good results (**Figure S7**). This is an additional evidence that the assembled nanotags notably improve the possibilities of using Au NSs for SERS bioimaging.

Figure 6. SERS maps and spectra from A549 cells incubated with 4-MBA labeled Au NS (**a**) and Au NS-PS beads (**b**). The concentration of Au NS was constant in both cases and SERS spectra were baseline corrected. SERS maps were created by plotting the intensity of vibration at 1078 cm⁻¹ (labeled with $*)$ as a function of the x,y position measured with step sizes of 2 μ m (**a**) and 1 μ m (**b**). The SERS spectra correspond to the points indicated with a white cross within each map. The intensity color scale for SERS spectra and 2D map in **a** was multiplied by 3 to create the SERS spectrum and map in **b**. White scale bars: 20 μm.

Cell viability assays were also carried out, for which A549 cells were incubated with different concentrations of Au NSs, Au NS-PS and PS particles for 66 h (**Figure S8**). Very low levels of toxicity were noted, mainly at the highest concentration, which - given the long incubation time - may actually be due to a reduction in cell proliferation rather than a toxic effect of the nanoparticle system.

We finally present a proof of concept experiment where the hybrid nature of the particles was exploited to build multifunctional nanotags for multimodal bio-imaging: Fluorescence SERS and Dark Field (DF) imaging. To this end, fluorescently labeled PS beads were used as colloidal support, so that complementary fluorescent imaging can be used to better understand relationship between cellular uptake, intracellular localization, and the recorded SERS signal. With a suitable strategy, a simple nanotag can be devised that enables us to independently monitor the nanoparticle, the supporting bead and the SERS label. Au NSs were thus assembled onto fluorescent PS beads, followed by functionalization with the 4- MBA SERS label. The dye in the PS beads was selected so as to avoid interference of the fluorescence (500-550 nm = green) with the excitation wavelength for SERS (785 nm = NIR). Additionally, the light scattering of nanoparticles depends on the material, size and shape,⁵⁹ which for Au NS allows detection under DF illumination conditions and application as a complementary cell imaging tool. The representative example shown in **Figure 7** allows us to see for the same A549 cells the strong SERS signal originating from 4-MBA-encoded Au NS-PS composites (**Figure 7a**), the strong scattered light from both the metal NPs and the PS cores **(Figure 7b**), as well as the bright green emission originating from the fluorophore in the polymer beads (**Figure 7c**). The highest SERS intensities in **Figure 7a** clearly correlate with the brightest areas in the dark field image (**Figure 7b**), whereas the z-stack measurements in **Figure 7c** also support the intracellular presence of the nanocomposites. This example is a clear evidence of the high versatility of this composite particle system to separately track and image cellular objects using correlated microscopy/spectroscopy tools. In a more realistic situation, the dual fluorescent-SERS tags could be labelled with several Raman reporter molecules and applied to multiplex biomarker detection.

Figure 7 Multimodal imaging of living A549 cells using *medium*-Au NS-PS beads via correlated optical spectroscopy. (**a**) Bright-field micrograph and 3D-intensity maps of the 1078 cm-1 vibration of 4-MBA in the SERS reporters, excited at 785 nm, for three different cells. The spatial resolution of the SERS images was 2 µm for 1 and 2, 1 µm for 3. (**b**) Dark field image of the same area. (**c**) Fluorescence mapping based on the emission (500-550 nm) from the fluorescently labeled PS beads. White scale bars: 20 μm.

CONCLUSIONS

A novel hybrid nanostructured system comprising Au NSs adsorbed on PS beads has been devised and fabricated, which shows great potential as multifunctional probe for bioimaging applications. Au NSs are highly efficient Raman-enhancing substrates which display LSPRs peaking in the biological transparency window. When assembled onto PS beads they present enhanced cell uptake, whereas the integration of various functionalities within the same composite particle allows us to image living cells using different analytical tools such as fluorescence, SERS and DF optical microscopy. Using 4-MBA as a Raman reporter molecule, the SERS efficiency of the hybrid composites containing different Au NS loadings was compared, obtaining for *medium* Au NS densities the highest SERS intensities. This intriguing finding was additionally supported by theoretical modeling and we expect that the results presented here will open up new insights toward engineering of Au NS-based plasmonic nanostructures.

Supporting Information

The Supporting Information is available free of charge on the ACS Publications website at DOI: 10.1021/acs.jpcb.xxxxxxx. More details on synthesis and assembly methods and characterization. Sample preparation and TEM and SERS images of particles in cells. Cell viability assays of cells after incubation with particles.

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