

This is the post-print version of the following article: *Mauri, M; Collico, V; Morelli, L; Das, P; Garcia, I; Penaranda, J; Bellini, M; Rotem, R; Truffi, M; Corsi, F; Simonutti, R; Liz-Marzán, LM; Colombo, M; Prospero, D. MnO Nanoparticles Embedded in Functional Polymers as T1 Contrast Agents for Magnetic Resonance Imaging ACS Applied Nano Materials 2020* DOI: [10.1021/acsnm.0c00474](https://doi.org/10.1021/acsnm.0c00474)

This article may be used for non-commercial purposes in accordance with ACS Terms and Conditions for Self-Archiving.

MnO Nanoparticles Embedded in Functional Polymers as T₁ Contrast Agents for Magnetic Resonance Imaging

Michele Mauri,^{1,†} Veronica Collico,^{2,†} Lucia Morelli,^{2,†} Pradip Das,² Isabel García,^{3,4} Jesus Penaranda Avila,² Michela Bellini,² Rany Rotem,² Marta Truffi,⁵ Fabio Corsi,^{5,6} Roberto Simonutti,¹ Luis M. Liz-Marzán,^{3,4,7} Miriam Colombo,^{2,} Davide Prospero^{2,6}*

¹ Department of Material Science, University of Milano-Bicocca, Via R. Cozzi 55, 20125, Milan, Italy. ² NanoBioLab, Department of Biotechnology and Biosciences, University of Milano-Bicocca, Piazza della Scienza 2, 20126, Milan, Italy. ³ CIC biomaGUNE, Basque Research and Technology Alliance (BRTA), Paseo de Miramón 182, 20014 Donostia–San Sebastián, Spain. ⁴ Centro de Investigación Biomédica en Red, Biomateriales, Bioingeniería y Nanomedicina (CIBER-BBN), Paseo de Miramón 182, 20014 Donostia–San Sebastián, Spain. ⁵ Nanomedicina Laboratory, Department of Biomedical and Clinical Sciences “L. Sacco”, via G. B. Grassi, 74, 20157 Milan, Italy. ⁶ Breast Unit, Surgery Department, ICS Maugeri IRCCS, via S. Maugeri 10, 27100 Pavia, Italy. ⁷ Ikerbasque, Basque Foundation for Science, 48013 Bilbao, Spain.

KEYWORDS: Manganese oxide nanoparticles; active polymer coating; magnetic resonance imaging; contrast agents; relaxivity; plasma effect.

ABSTRACT

The design and development of contrast agents (CAs) for magnetic resonance imaging (MRI) in clinical analysis is expected to improve image spatial resolution and to increase detection sensitivity, especially regarding neurological disorders and cancer disease. In particular, advanced CAs for T₁-weighted images are investigated to achieve sensitive detection of early stage primary tumors or brain metastases. In this study, we present a strategy toward diagnostic T₁ contrast agents for MRI, based on polymer-modified MnO nanoparticles. Two different nanosystems were synthesized, consisting of 1) colloidal MnO nanocrystals wrapped by a multidentate amphiphilic polymer, and 2) MnO nanocrystals embedded into PLGA nanoparticles. These nanosystems were compared in terms of their MRI contrast power and biological safety. The latter combines the excellent biocompatibility of PLGA with the unique magnetic properties of MnO nanoparticles and allows sustained contrast enhancement over time. Longitudinal relaxivities of both MnO composite nanomaterials proved to be higher than those of commercial Gd-based contrast agents and Teslascan[®], both in phosphate buffer saline and in plasma, also exhibiting low cytotoxicity. The high relaxation rates achieved with these contrast enhancers are promising toward future application in *in vivo* imaging.

INTRODUCTION

Magnetic resonance imaging (MRI) is a non-invasive and efficient medical imaging technique used in daily clinics to visualize the anatomy and physiological processes of patients in real-time.¹ Contrast agents (CAs) are often used in MRI, enhancing image spatial resolution and increasing detection sensitivity.² CA efficiency is measured by their longitudinal (r₁) and transverse (r₂) molar

1
2
3 relaxivities, associated to their capability to shorten the spin-lattice (T_1) and spin-spin (T_2) nuclear
4 relaxation times of water protons. CA relaxivity can be tuned by several methods, resulting in the
5 design of a great diversity of agents.³ Positive CAs (high r_1) are mainly used for neurological
6 examinations and diagnosis of neurodegenerative disorders.^{4,5} T_1 -weighted images are also
7 expected to allow the direct detection of early primary tumors or brain metastasis.⁶ In the clinics,
8 the most frequently used CAs are gadolinium-based probes.^{7,8} However, despite their worldwide
9 application, conventional Gd-chelated ion probes exhibit two major limitations: poor sensitivity
10 and, in some cases, long-term accumulation and nephrotoxicity.⁹ Although gadolinium remains
11 the gold standard CA in clinical MRI, several cases of nephrogenic systemic fibrosis have been
12 reported in patients treated with Gd CAs^{10,11,12,13,14} and, as a result, four Gd-based CAs have been
13 withdrawn from the market in 2017.¹⁵ Such drawbacks reduced the availability of positive CAs,
14 especially if compared with the variety of T_2 -based CAs, spawned by the exploitation of
15 superparamagnetic iron oxide nanoparticles¹⁶ or fluorinated copolymers.¹⁷ In this context,
16 manganese ions and their derivatives have been explored as promising alternatives to positive Gd
17 CAs. Mn^{2+} ions have demonstrated a T_1 contrast power comparable to that of Gd(III), increasing
18 the signal intensity of T_1 -weighted images.¹⁸ A great advantage of using Mn^{2+} resides in its lower
19 toxicity compared to Gd^{3+} at an equivalent dose.¹⁹ Indeed, manganese is a natural cellular
20 constituent and therefore often found acting as a protein cofactor.^{20,21} Hitherto, examples of Mn(II)
21 CAs mainly consist of soluble complexes,^{22,23} while only a few Mn CAs based on manganese
22 oxide (MnO) nanomaterials,^{24,25,26} metal-organic frameworks²⁷ and manganese-oxo clusters have
23 been reported.^{28,29} Mangafodipir (trisodium Mn(II)-*N,N'*-dipyridoxyl ethylenediamine-*N,N'*-
24 diacetate-5,50-bisphosphate) was the first Mn CA initially approved by the FDA and the EMA, to
25 be commercialized as Teslascan for hepatocyte detection.³⁰ Unfortunately, Teslascan© was
26
27
28
29
30
31
32
33
34
35
36
37
38
39
40
41
42
43
44
45
46
47
48
49
50
51
52
53
54
55
56
57
58
59
60

1
2
3 withdrawn from the market in June 2012 for demonstrated neurotoxicity and commercial
4 reasons,^{31,32} thus there are no presently available Mn CAs. The development of molecular
5 formulations is progressing, in particular through the rational synthesis and evaluation of libraries
6 of variant molecules,³³ but a key step forward in expressing the diagnostic potential of Mn CAs is
7 expected by the contribution of nanotechnology. The ability to rationally manipulate nanoparticle
8 (NP) features such as size, shape and surface coating, opens up new opportunities in designing
9 diagnostic nanocarriers to selectively target cells, tissues and metabolic processes.^{34,35}

10
11
12
13
14
15
16
17
18
19 In most studies, MnO particles exhibit very low molar relaxivities, usually in the range 0.2-0.5
20 mM⁻¹ s⁻¹, at least one order of magnitude lower than free Mn²⁺ ions at the same manganese
21 concentration.³⁶ Such small contrast enhancement in MRI has been attributed to the lack of contact
22 between the core Mn atoms and the surrounding water molecules. Recently, 15-20 nm nanoclusters
23 incorporated inside 140 nm and 1.7 μm poly(lactic-*co*-glycolic acid) (PLGA) particles were
24 reported to strongly enhance the MRI contrast under acidic environment, owing to Mn²⁺ release
25 from the polymer matrix.³⁷ However, the effectiveness of Mn²⁺ ions is strongly dependent on the
26 interaction with cellular membranes, making the effectiveness of preparations reliant on single Mn
27 ions difficult to validate.³⁸ On the other side, there is evidence that the interaction of the MnO
28 particle as a whole can be more specifically effective than that of Mn ions: for example, differently
29 shaped particles can display starkly different relaxivities.³⁹ Also, nanoparticles have the flexibility
30 to be deployed as tunable or activatable systems.⁴⁰ Recent work indicates that the relaxivity is
31 extremely sensitive, not only to the particle itself, to the point that it can even be used to distinguish
32 the surface energy of diamagnetic nanoparticles,⁴¹ but also to the coating, since it modulates the
33 interactions between water and the NP surface.⁴² In the present work, two different kinds of MnO
34 NPs, based on the same cores but with different polymer coatings, were investigated as safe
35
36
37
38
39
40
41
42
43
44
45
46
47
48
49
50
51
52
53
54
55
56
57
58
59
60

1
2
3 positive CAs with improved MRI performance. One system consists of nanoparticles individually
4 coated with the amphiphilic multidentate polymer poly(isobutylene-*alt*-maleic anhydride), or
5 PMA, which provides an individual coverage to each nanoparticle. The other system comprises
6 multiple particles embedded into large nanoparticles of poly(D,L-lactide-*co*-glycolide) (PLGA
7 NPs).
8
9
10
11
12
13
14
15
16

17 EXPERIMENTAL SECTION

18
19 **Materials.** Manganese chloride tetrahydrate ($\text{MnCl}_2 \cdot 4\text{H}_2\text{O}$, 99%), sodium oleate (82%), oleic
20 acid (analytical standard), poly(dl-lactide-*co*-glycolide) (PLGA, MW 24-38 kDa, 50:50), and
21 polyvinyl alcohol (PVA, MW 9-10 kDa, 80% hydrolyzed) were purchased from Sigma-Aldrich
22 (MO, USA). Hexane (97%), chloroform and acetone (99%) were acquired from PanReac
23 AppliChem.
24
25
26
27
28
29

30
31 **Synthesis of PMA coated manganese oxide nanoparticles (MnO@PMA NPs).** The synthesis
32 of oleic acid-coated manganese oxide nanoparticles (MnO NPs) was performed according to a
33 previously reported method.⁴³ In brief, the metal precursor (manganese oleate) was synthesized
34 *via* overnight refluxing a water-ethanol-hexane solution of manganese (II) salt and sodium oleate.
35 Next, the thermal decomposition of pure manganese oleate in a high boiling solvent, 1-ocadecene,
36 produced high quality hydrophobic MnO NPs.^{28, 43} The so-prepared MnO NPs were subsequently
37 transferred to an aqueous phase by polymeric coating with poly[isobutene-*alt*-maleic anhydride]
38 reacted in organic solvent with dodecylamine in a 4:3 monomer to dodecylamine molar ratio
39 (PMA).⁴⁴
40
41
42
43
44
45
46
47
48
49
50

51
52 **Synthesis of poly(D,L-lactide-*co*-glycolide) nanoparticles (PLGA NPs).** The carboxyl group-
53 terminated poly(D,L-lactide-*co*-glycolide) (MW 24-38 kDa) nanoparticles (PLGA NPs) were
54
55
56
57
58
59
60

1
2
3 synthesized by the single emulsion solvent evaporation method.⁴⁵ Briefly, 12.5 mg of PLGA
4 dissolved in chloroform (1 mL) was added to an aqueous solution of polyvinyl alcohol (PVA, MW
5 9-19 kDa, 40 mL, 2% w/w), to form an emulsion under ultrasonic stirring at 40% of amplitude for
6 30 seconds (twice) in an ice-bath (tip-sonicator Branson, Digital Sonifier). The emulsion was left
7 under magnetic stirring to completely evaporate the organic solvent (4 h). The sample was purified
8 by three-fold centrifugation at 27,237 ref for 15 min at 4 °C (Scanspeed 173OR, Labogene) and
9 the pellet was re-dispersed twice with PVA solution (0.3 % w/w). Finally, the sample was freeze-
10 dried (Christ, alpha 1-2 LD).

11
12
13 **Synthesis of poly(D,L-lactide-co-glycolide) coated manganese oxide nanoparticles**
14 **(MnO@PLGA NPs).** The carboxyl group-terminated poly(D,L-lactide-co-glycolide) (MW 24-38
15 kDa) nanoparticles containing MnO NPs (MnO@PLGA NPs) were synthesized through adding 8
16 mg of MnO NPs to an organic solution of PLGA before ultrasonic stirring under the same
17 experimental conditions. Further steps as mentioned above for the synthesis of PLGA NPs were
18 followed to entrap multiple MnO nanoparticles inside PLGA NPs.

19
20
21 **Nanoparticles characterization by DLS and ζ -potential.** NP size was characterized by
22 dynamic light scattering (DLS) using a particle size analyzer (90Plus/BI-MAS, Brookhaven
23 Instruments Corp.) operating at 4 mW of a solid-state laser ($\lambda = 633$ nm) with a scattering angle
24 of 173°. Measurements were carried out at 25 °C using a disposable cuvette with 10 mm optical
25 path length. Three measurements per sample were performed to test the repeatability (120 μ L, 10
26 μ g mL⁻¹ for metal oxide samples and 25 μ g mL⁻¹ for polymeric samples). Mean hydrodynamic
27 diameters and standard deviations (SD) were calculated with multimodal analysis using Mie
28 theory, by considering the absolute viscosity and the refractive index values of the aqueous
29 medium to be 0.8872 cP and 1.330, respectively. Zeta-potential (ζ) was determined with the same
30
31
32
33
34
35
36
37
38
39
40
41
42
43
44
45
46
47
48
49
50
51
52
53
54
55
56
57
58
59
60

1
2
3 instrument, equipped with a DTS1070 electrode and data were processed by ZetaPlus Software. ζ -
4
5 p was automatically calculated from the electrophoretic mobility, based on Smoluchowski theory.
6
7 A viscosity value of 0.8872 cP, a dielectric constant of 78.5, and Henry function of 1.5 were used
8
9 for calculations. The mean value from ten measurements gives one record. Analyses were
10
11 performed in triplicate for each sample (1 mL, 1 $\mu\text{g mL}^{-1}$ for metal oxide samples and 3 $\mu\text{g mL}^{-1}$
12
13 for polymeric samples).
14
15

16
17 **Morphological characterization by TEM.** NPs morphology was investigated by transmission
18
19 electron microscopy (FEI Morgagni at 80 kV) and digital images were obtained by means of a
20
21 CCD Camera System and the Leo Image software. Samples (0.05 $\mu\text{g mL}^{-1}$) were placed onto a
22
23 formvar-coated 300 mesh copper grid (Ted Pella, Inc.). The specimen on the copper grid was
24
25 negatively stained with uranyl acetate (0.2% w/w) only for polymeric samples. Measure IT
26
27 Olympus Software was used to determine the particle size distribution by measuring about 150
28
29 particles (image processing).
30
31

32
33 **Inductively Coupled Plasma–Optical Emission Spectrometry (ICP–OES) analysis.** Mn
34
35 contents were determined by ICP-OES (ICAP 6300, Thermo Fisher Scientific) and an external
36
37 calibration methodology, after sample degradation in a mixture of 0.75 mL of HCl (36 vol%) and
38
39 0.25 mL of HNO_3 (68 % v/v), diluted to 4 mL with Milli-Q water overnight.
40
41

42 **Mn^{2+} encapsulation efficiency (%) and loading ($\mu\text{g mg}^{-1}$).** Encapsulation efficiency expresses
43
44 the amount of metal oxide nanoparticles loaded inside PLGA matrix after NP formulation and was
45
46 calculated as reported in Eq. 1. Mn^{2+} loading ($\mu\text{g mg}^{-1}$) is a parameter related to the amount of
47
48 NPs recovered by ICP analysis after the formulation and the theoretical yield (Eq. 2). ICP analysis
49
50 were performed by ICP-OES.
51
52

53
54 Equation (1):
55
56
57
58
59
60

$$\text{encapsulation efficiency}(\%) = \text{mass of } \frac{Mn^{2+} (\mu g)_{\text{entrapped}}}{\text{mass of } Mn^{2+} (\mu g)_{\text{added}} \times \text{reaction}} * 100$$

Equation (2):

$$\text{loading}(\mu\text{gmg}^{-1}) = \text{mass of } \frac{Mn^{2+} (\mu\text{g})}{\text{mass of PLGANPs}(\text{mg})}$$

Relaxometric analysis. The magnetic properties of MnO@PLGA NPs and other particles were characterized with a 0.5 T (20 MHz) Time-Domain (TD) NMR benchtop system (Bruker Minispec mq20) with N₂ heating of the sample region. In a typical experiment, 120 μL of NP dispersion was analyzed in 10 mm diameter NMR glass tubes, filling the bottom 8 mm of depth. Standard CPMG (cp_mb) and saturation recovery (sr_mb) sequences were applied to determine the relaxation times T₂ and T₁, respectively. Measurements were performed at 303 K in different media (saline and plasma mixture). Plasma samples were freshly collected from healthy volunteers. In fact, the nanoparticles cannot be prepared directly in plasma, since after preparation they are dispersed in water and cannot be completely lyophilized without altering them. So, diluted stable water dispersion with the maximum possible concentration with plasma, preparing 58.3% v/v solutions. R₁ and R₂ relaxation rates (R) were calculated as 1000/T₁ or T₂ respectively, while r₁ and r₂ relaxivities (r) coincide with the angular coefficient of the calibration curve R vs. concentration, obtained from up to 6 successive dilutions. CPMG experiments were repeated twice on each sample: once before and once after the longer saturation recovery experiment. This scheme ensures the detection of any rapid change in relaxivity, due for example to particle precipitation and aggregation.

In vitro MRI analysis. The magnetic property of MnO@PLGA NPs was characterized by MRI analysis using a 7 T small-bore magnetic resonance scanner equipped with 450/675 mT/m

1
2
3 gradients (slew-rate: 3,400-4,500 T/m/s; rise-time 140 μ s) and a mouse body volume-coil
4 (BRUKER, BioSpec 70/30 USR, Paravision 5.1). 500 μ L of samples at different concentrations
5 (from 0.003 to 1.0 mM Mn(II) ion concentration) were analyzed at 20.6 $^{\circ}$ C \pm 0.1 $^{\circ}$ C. RAREVTR
6 (Rapid Acquisition with Relaxation Enhancement (RARE) with variable repetition time) and Multi
7 Slice Multi Echo (MSME) sequences were applied to determine T_1 and T_2 , respectively.
8 Measurements were performed at 303 K in saline (0.9% NaCl) or in mixtures of saline and plasma
9 (58.3% v/v). Relaxation rates R and relaxivities were calculated as described above. The software
10 processes the T_1 -map and provides the MR images, in which the dynamic contrast is shown as a
11 bright signal.
12
13
14
15
16
17
18
19
20
21
22
23

24 **Cell culture.** HeLa cervical cancer cell line was used as *in vitro* cell culture model. Cells were
25 cultured in high glucose Dulbecco's modified Eagle medium (DMEM) supplemented with 10%
26 heat inactivated fetal bovine serum (FBS), 2 mM l-glutamine, 100 U mL⁻¹ penicillin, 0.1 mg mL⁻¹
27 streptomycin at 37 $^{\circ}$ C in a humidified atmosphere containing 5% CO₂. At confluence, the cells
28 were treated using trypsin-EDTA in a 1:4 to 1:6 split ratio, or they were used for experiments. All
29 cell culture materials were supplied by Euroclone S.p.A. (Milano, IT).
30
31
32
33
34
35
36
37

38 **MTT assay.** In vitro cytotoxicity of MnO@PMA, MnO@PLGA and PLGA NPs was tested on
39 HeLa cells through an MTT assay (CellTiter 96 Non-Radioactive Cell Proliferation Assay,
40 Promega). Cells were seeded in a 96-well plate at a density of 5×10^3 cells per well and incubated
41 for 24 h. Then, cells were incubated with different amounts of the NPs to be tested, six replicates
42 per each concentration. According to the manufacturer's instructions, at the end of the exposure
43 time (24, 48 and 72 h), MTT was added and formazan product was detected after 4 h at 37 $^{\circ}$ C
44 reading the absorbance at 570 nm with EnSight™ Multimode Plate Reader (Perkin Elmer)
45
46
47
48
49
50
51
52
53
54
55
56
57
58
59
60

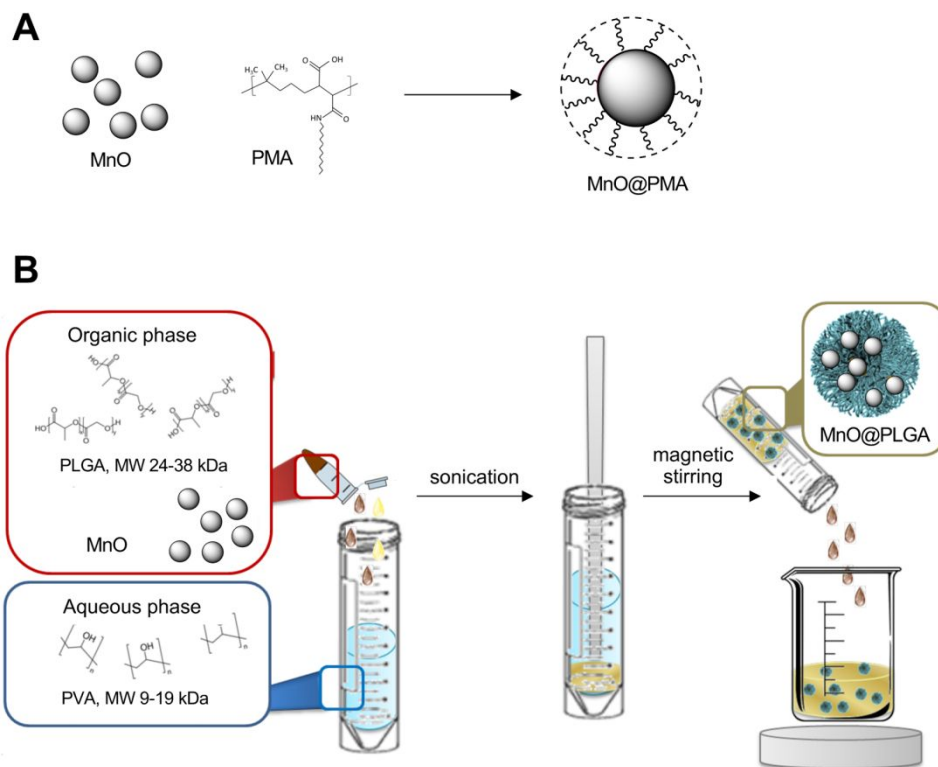
1
2
3 subtracting the absorbance of background at 620 nm. Results were obtained by normalizing the
4 untreated cells set at 100% and expressed as percentage \pm SD.
5
6

7 **XRD:** For X-ray characterization, a D8 Advance powder diffractometer (Bruker) was used with
8 Cu K α 1 radiation ($\lambda = 1.5418 \text{ \AA}$) and secondary-beam monochromator. The powder is added on a
9 quartz sample holder and measured.
10
11
12
13
14
15
16

17 RESULTS AND DISCUSSION

18 **Synthesis and chemical-physical characterization of MnO@PLGA and MnO@PMA NPs.**

19 A schematic representation of the preparation of MnO@PLGA and MnO@PMA NPs is illustrated
20 in Scheme 1. In a first setting, a biodegradable PLGA polymer matrix was used to trap and carry
21 MnO NPs, pre-synthesized in organic solvent as MnO nanocrystals stabilized by oleic acid coating,
22 obtaining MnO@PLGA NPs (Scheme 1B). This strategy was aimed at concentrating a large
23 number of MnO NPs inside a nanocarrier that could allow the simultaneous delivery of a sufficient
24 amount of Mn, to cause localized increase in T₁ relaxivity, while decreasing the metal toxicity
25 associated to free ionic Mn²⁺. The aim of this study was to compare the magnetic properties as T₁
26 CAs and the biological effects of MnO@PLGA NPs with those of colloiddally stable MnO NPs
27 that were obtained by individually wrapping the same paramagnetic MnO nanocrystals into an
28 amphiphilic multidentate polymer poly(isobutylene-alt-maleic anhydride), or PMA, endowed with
29 high affinity for hydrophobic particles (Scheme 1A).
30
31
32
33
34
35
36
37
38
39
40
41
42
43
44
45
46
47
48
49
50
51
52
53
54
55
56
57
58
59
60



Scheme 1. Schematic representation of the synthesis of a) MnO@PMA and b) MnO@PLGA NPs.

As-synthesized, surfactant-coated MnO nanocrystals^{28, 43} showed a hydrodynamic diameter of 15.9 ± 0.2 nm in chloroform, as measured by dynamic light scattering (DLS, Fig. 1A), in agreement with transmission electron microscopy (TEM, Fig. 1B), which exhibited homogeneous quasi-spherical NPs of 9.7 ± 1.3 nm. Water-soluble MnO (MnO@PMA) NPs with high colloidal stability were obtained by coating MnO nanocrystals with PMA, synthesized by condensing poly(isobutylene-*alt*-maleic) anhydride with dodecylamine, as previously reported.⁴⁴ MnO@PMA NPs exhibited a hydrodynamic diameter of 17.7 ± 0.8 nm (Fig. S1E in the Supplemental Material), and a diffractogram typical of MnO nanocrystals covered with a relatively thick layer of amorphous polymer (Fig. S2).⁴⁶ PLGA NPs were prepared by a single emulsion method (see Methods section).⁴⁵ To obtain MnO@PLGA NPs, MnO nanocrystals were added to a PLGA (MW 24-38 kDa) solution in chloroform, mixed with an aqueous solution of 2% polyvinyl

1
2
3 alcohol (MW 9-19 kDa), sonicated and freeze dried. Both empty PLGA NPs and MnO@PLGA
4 NPs were colloiddally stable in aqueous media and showed comparable sizes (150.5 ± 8.0 nm and
5
6
7
8 163.0 ± 3.3 nm, respectively, Fig. 1A), with negative surface charge (ζ -potential of -33.5 ± 0.35
9
10
11
12
13
14
15
16
17
18
19
20
21
22
23
24
25
26
27
28
29
30
31
32
33
34
35
36
37
38
39
40
41
42
43
44
45
46
47
48
49
50
51
52
53
54
55
56
57
58
59
60

alcohol (MW 9-19 kDa), sonicated and freeze dried. Both empty PLGA NPs and MnO@PLGA NPs were colloiddally stable in aqueous media and showed comparable sizes (150.5 ± 8.0 nm and 163.0 ± 3.3 nm, respectively, Fig. 1A), with negative surface charge (ζ -potential of -33.5 ± 0.35 mV and -32.1 ± 0.99 mV, respectively). Data from TEM were consistent with DLS size and showed that the NPs maintained their spherical shape after MnO loading (Fig. 1C,D and S1). Figure 1D clearly shows that several MnO nanocrystals were included in each PLGA polymer NP. Notably, the nanocrystals were preferentially located inside the inner core, rather than at the outer shell. Both DLS and TEM size distributions suggest a high degree of monodispersity (Table 1, Fig. 1 and Fig. S1). We thus concluded that the encapsulation of metal oxide nanoparticles did not significantly affect the overall shape, dimension and charge of PLGA NPs.

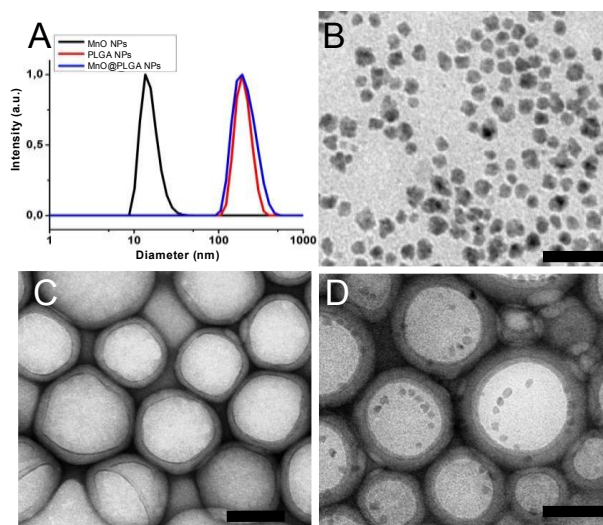


Figure 1. A) Hydrodynamic diameter of MnO, PLGA and MnO@PLGA NPs measured by DLS. B-D) TEM images of (B) MnO (Scale bar = 50 nm), (C) PLGA and (D) MnO@PLGA NPs (Scale bars = 100 nm). Samples were negatively stained with uranyl acetate (0.2% w/w).

Table 1. DLS and ζ -potential data of MnO NPs, PLGA NPs and MnO@PLGA NPs.

	TEM diameter (nm)	DLS diameter (nm)	PDI	ζ (mV)
MnO NPs	9.7 ± 1.3	15.9 ± 0.2	0.130 ± 0.034	—
MnO@PMA	$9.7 \pm 1.3^*$	17.7 ± 0.8	0.296 ± 0.22	-40.8 ± 0.70
PLGA NPs	129.5 ± 35.1	150.5 ± 8.0	0.058 ± 0.028	-33.5 ± 0.3
MnO@PLGA NPs	132.8 ± 41.9	163.0 ± 3.3	0.117 ± 0.013	-32.1 ± 0.99

* same as MnO NPs, PMA not visible at TEM

We assessed the colloidal stability of MnO@PMA and MnO@PLGA NPs in physiological phosphate buffer saline (PBS, pH 7.2) to estimate their potential for biomedical applications. MnO@PMA NPs were highly stable in water and PBS for at least 10 days (Fig. 2A), after an initial swelling from the size values indicated in Table 1 to slightly higher ones, closer to 200 nm. This difference between the first measurable value and the equilibrium value is in line with the smaller diameters measured by TEM, which were performed in vacuum and therefore do not reflect any swelling. MnO@PLGA NPs were first lyophilized for storage, showing a stability of at least 10 days after reconstitution in PBS, according to DLS data (Fig. 2B). An MnO loading of 13.8 ± 2.2 μg per mg of PLGA, corresponding to an encapsulation efficiency (EE%) of $9.65 \pm 1.31\%$, was determined by Inductively Coupled Plasma – Optical Emission Spectrometry (ICP-OES) analysis and proved to be reproducible from batch to batch. For MnO@PMA, a MnO loading of 36 μg per mg of PMA was obtained.

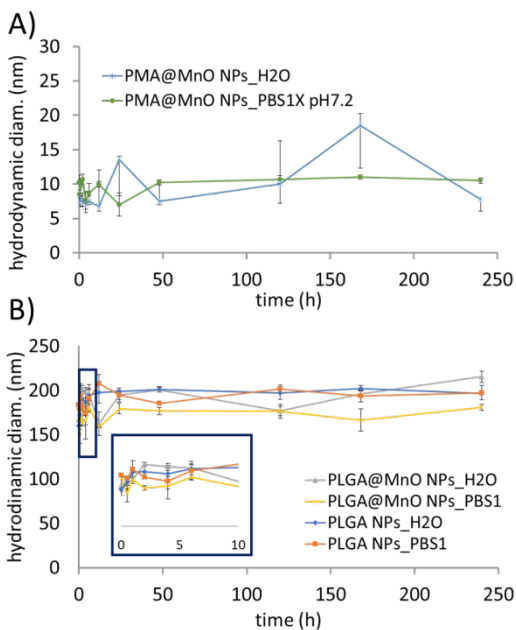


Figure 2. Colloidal stability of (A) MnO@PMA NPs, (B) PLGA NPs and lyophilized MnO@PLGA NPs in water and phosphate buffer solution over 10 days, measured by DLS analysis. The inset shows the evolution during the initial 10 hours.

Relaxometric determination of contrast power of MnO@PLGA and MnO@PMA NPs in aqueous buffer and human plasma. Relaxivity (r) measurements for MnO@PMA and MnO@PLGA NPs were conducted at 310 K, to evaluate the ability of Mn-bearing NPs to reduce the relaxation rates (R) of water protons at 0.5 T, at a temperature close to real biosystems. The longitudinal (T_1) and transverse (T_2) relaxation times of MnO@PMA and MnO@PLGA NPs were first measured in phosphate buffer solution (PBS) with the typical sequences (CPMG and saturation recovery) used for the characterization of magnetic NPs⁴⁷. MnO@PMA NP relaxation rates (R_1 and R_2 , calculated as $1/T_1$ and $1/T_2$, respectively, plotted in Fig. 3A) were found to linearly depend on CA concentration, within a wide range (up to the maximum obtainable Mn^{2+} concentration around 0.5 mM), thereby confirming NP stability and absence of aggregation. The longitudinal and transverse relaxivities (r_1 and r_2 , determined as the slope in the R vs. $[Mn^{2+}]$ plot,

see Fig. 3) in water were estimated to be $7.7 \text{ mM}^{-1} \text{ s}^{-1}$ and $10.5 \text{ mM}^{-1} \text{ s}^{-1}$, respectively (Fig. 3A). The r_2/r_1 ratio is an important parameter to estimate whether a contrast enhancer can serve as either T_1 or T_2 CA. T_1 -agents, such as paramagnetic chelates, usually exhibit r_2/r_1 ratios in a range of 1 to 2.⁴⁸ $\text{MnO}@PMA$ NPs are thus well suited as T_1 CA, having $r_2/r_1 = 1.2$ and high r_1 absolute relaxivity values, in line with reportedly excellent NPs.⁴⁹ In addition, $\text{MnO}@PMA$ NPs maintained constant relaxivity values in water for at least 24 h (Fig S3). Fig. 3B indicates that the linearity of relaxation rate with concentration was maintained for $\text{MnO}@PLGA$ in PBS, up to a concentration of 11 mM. Encapsulation of MnO nanocrystals into PLGA as $\text{MnO}@PLGA$ NPs increased the solubility, and thus the maximum obtainable concentration of Mn^{2+} in the dispersion increases to 11 mM.

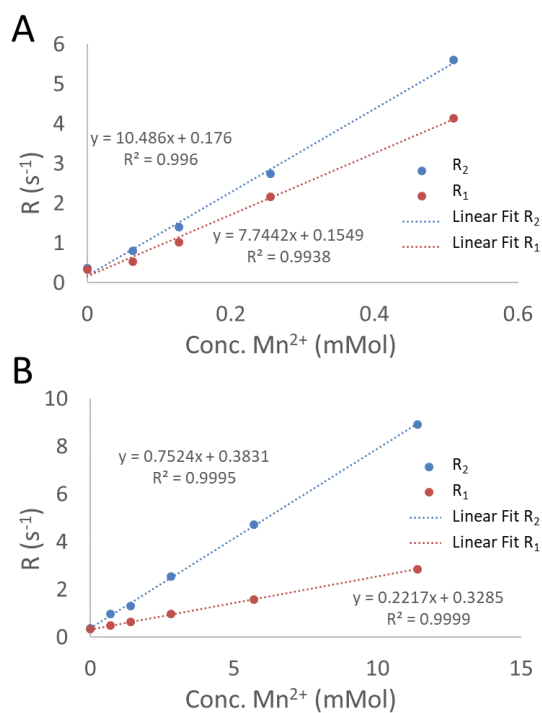


Figure 3. r_1 and r_2 relaxivities calculated from measuring the relaxation rate at different dilutions of A) $\text{MnO}@PMA$ in PBS (R_2 : $y = 10.5x + 0.2$, $R^2 = 0.996$; R_1 : $y = 7.74x + 0.2$, $R^2 = 0.994$; B) $\text{MnO}@PLGA$ NPs in PBS (R_2 : $y = 0.75x + 0.4$, $R^2 = 0.9995$; R_1 : $y = 0.22x + 0.3$, $R^2 = 0.9999$)

1
2
3 Interestingly, the absolute values of relaxation rates are not much higher than those achievable
4 by PMA coated nanoparticles at lower concentrations, resulting in much smaller values of r_1 and
5 r_2 (r_1 : $0.22 \text{ mM}^{-1} \text{ s}^{-1}$; r_2 : $0.75 \text{ mM}^{-1} \text{ s}^{-1}$; $r_2/r_1 = 3.4$, Fig. 3C). This variation of more than one order
6 of magnitude is due to the presence of the PLGA matrix in MnO@PLGA NPs, which is likely to
7 reduce the CA interaction with water molecules, thereby decreasing the spin-spin and spin-lattice
8 relaxation from the CA to the surrounding medium much more pronouncedly, as compared to the
9 thin PMA layer in MnO@PMA NPs.⁵⁰ The detected 3-fold increase of the r_2/r_1 ratio, in comparison
10 to MnO@PMA NPs also points toward a difference in accessibility, since T_2 relaxation is related
11 to longer range effects compared to T_1 , and the ratio changes since r_1 was more strongly affected
12 by PLGA encapsulation. Importantly, also MnO@PLGA NPs relaxivities remained constant over
13 24 h in water (Fig. S3). To evaluate the contrast power of the NPs, their longitudinal relaxivity
14 was then compared with Teslascan© data available in the literature (r_1 : $1.85 \text{ mM}^{-1} \text{ s}^{-1}$; r_2 : 2.18
15 $\text{mM}^{-1} \text{ s}^{-1}$; $r_2/r_1 = 1.2$).⁵¹ The relaxation of MnO@PMA was found to be even higher than that
16 reported for Teslascan©, suggesting a superior efficacy for MnO@PMA NPs as a contrast
17 enhancer *in vitro*, compared to commercial CAs. Note that this is an improvement also over many
18 Mn systems presented in literature, that mostly present lower absolute r_1 values,²⁶ or very high r_2/r_1
19 ratios.²⁷ In contrast, the relaxation of MnO@PLGA NPs was 2.7 times lower than that of
20 Teslascan©. However, in order to fully appreciate the potential of these composite NPs for future
21 *in vivo* MRI applications, relaxation times for both MnO@PMA and MnO@PLGA NPs were
22 further analyzed in human plasma (Figs. 4 and 5). For the experiments in Fig, 4, we prepared
23 solutions with different amounts of MnO particles, selected to start with roughly the same absolute
24 R_1 .

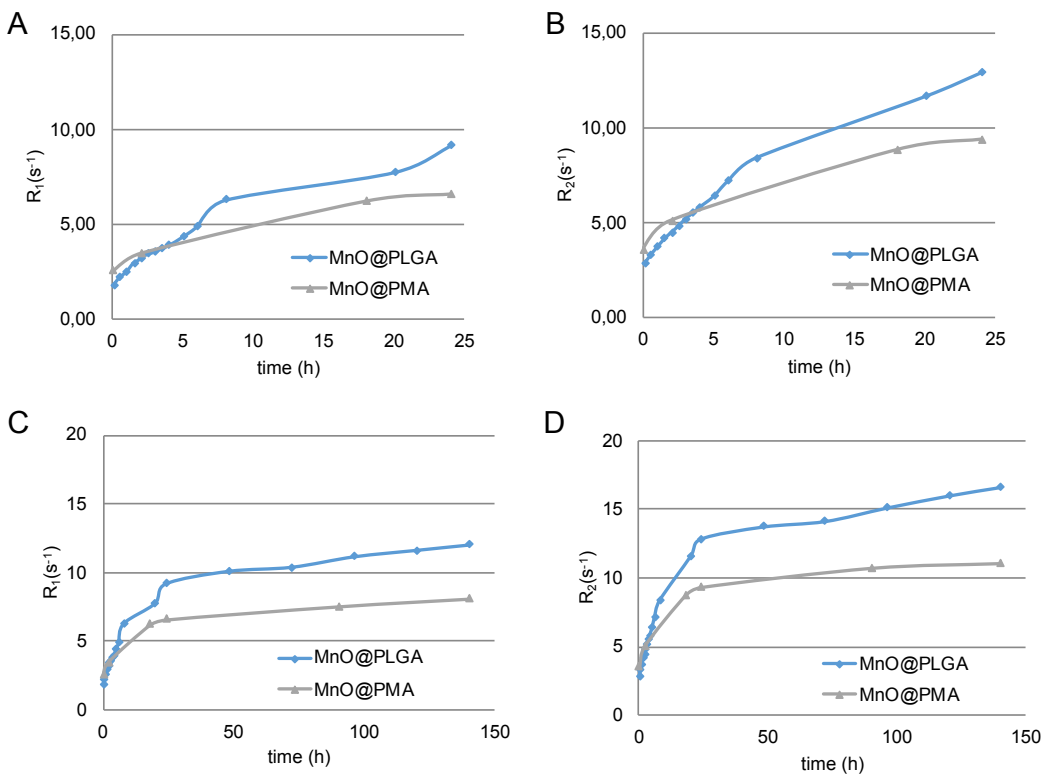


Figure 4. R_1 and R_2 of MnO@PLGA and MnO@PMA NPs ($120 \mu\text{L}$, $917 \mu\text{g mL}^{-1}$ MnO@PLGA NPs, $88 \mu\text{g mL}^{-1}$ MnO@PLGA) relaxation rates in plasma from three different donors for 24 h (A and B, respectively) and for 6 days (C and D, respectively).

In plasma, MnO@PMA r_1 and r_2 increased immediately after dispersion to $14.5 \text{ mM}^{-1} \text{ s}^{-1}$ and $16 \text{ mM}^{-1} \text{ s}^{-1}$, respectively, whereas the r_2/r_1 ratio remained near 1.1. For MnO@PLGA, relaxivities measured immediately after sample preparation were $0.68 \text{ mM}^{-1} \text{ s}^{-1}$ (r_1) and $0.71 \text{ mM}^{-1} \text{ s}^{-1}$ (r_2) (Fig. 3S B,C). This also brought the ratio close to 1, indicating a different mechanism of interaction, which is to be expected since the interaction between the NPs embedded in the PLGA coating and water is mediated by PLGA itself, which is sensitive to the presence of various plasma molecules. Still, the absolute values were low, as observed when assessing Gd response in plasma, where gadolinium forms complexes with blood proteins which partially mask the paramagnetic effect.¹⁹

Both MnO@PMA and MnO@PLGA NPs showed kinetic profiles in plasma that were significantly different from those measured in PBS (Fig. 4). In plasma, we observed a burst increase in both R_1 and R_2 within the first 6 h, followed by a change in slope leading to a milder increase during the subsequent 24 h. This experiment was extended for 120 h, confirming the trend.

Since the effect appeared much more pronounced in MnO@PLGA, with an increase of one order of magnitude, we replotted the data from plasma and PBS in Fig. 5, also reporting the r_2 . The wide error bars present in the figure indicate a high sensitivity to the specific composition of the plasma from different donors. Still, the kinetic profile was consistently the same, with both relaxation values surpassing those of Teslascan© over time. This confirms that the increased relaxation properties are inherent to the evolution of the material and independent from inhomogeneities of the environment composition. An estimate of r_1 and r_2 , based on the ratio between maximum R values and concentration, yielded $11.38 \text{ mM}^{-1} \text{ s}^{-1}$ for r_2 and $8.26 \text{ mM}^{-1} \text{ s}^{-1}$ for r_1 . We assumed that the increase of R_1 and R_2 over time was due to a release of either Mn^{2+} ions or MnO NPs from the carriers, resulting in a more effective interaction with the surrounding medium. The values estimated above also produced an r_2/r_1 ratio of 1.38, indicating that the most relevant mechanism is the interaction of NPs with water.

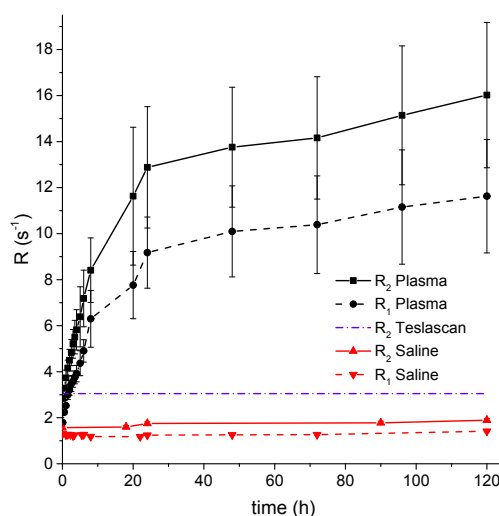


Figure 5. R_1 and R_2 values for MnO@PLGA in PBS and in plasma, up to 120 h. Both samples contained 100 mg mL^{-1} of CA, corresponding to 1.4 mmol of Mn^{2+} . The dashed line represents the expected R_2 relaxivity for the same amount of Teslascan $^{\text{®}}$. The values presented for plasma are an average from three experiments performed with plasma from different donors. SDs for measurements conducted in PBS were $<1\%$.

The phenomena discussed above are summarized in Table 2, where both the high intrinsic efficiency of MnO@PMA NPs and the peculiar behavior of MnO@PLGA in plasma clearly stand out.

Table 2. Summary of r_1 and r_2 values of the novel NPs and the Teslascan $^{\text{®}}$ benchmark. Relaxivities are expressed in $\text{mM}^{-1} \text{ s}^{-1}$. Values in *italic* are not rigorously obtained as a slope, but rather estimated using a single point at a known concentration. Standard deviations omitted for clarity are less than 4% except for MnO@PLGA, where differences between plasma donors increase the variability.

Sample	r_1	r_2	r_2/r_1	Time Evolution r_1
In PBS				
MnO@PMA	7.7	10.4	1.35	Stable
MnO@PLGA	0.24	0.82	3.4	Stable
Teslascan $^{\text{®}}$	1.85	2.18	1.2	Stable
In Plasma				
MnO@PMA	<i>14.5</i>	<i>16.0</i>	1.1	Increase to 31 ± 1
MnO@PLGA	<i>0.68</i>	<i>0.71</i>	1.05	Increase to 8.3 ± 1.5

Possible mechanisms for the latter effect include the swelling of the polymeric matrix or the faster degradation of PLGA polymer in plasma compared to PBS. Indeed, released MnO NPs were expected to produce also a progressive and sustained release of Mn²⁺ ions, with a concomitant increase in T₁-weighed MR signal. It is likely that the small aqueous CAs exhibited higher relaxation rates by reducing molecular motion due to interactions with serum proteins.⁵²⁵³ In support to this interpretation, we observed that Mn²⁺ ions in plasma led to greater T₁ shortening as compared to PBS, in a wide concentration range (Table 3). On the other hand, the ions produced by dissolution of MnCl₂ did also produce a very high r₂/r₁ ratio, around 10 in PBS and 5 in plasma, supporting the hypothesis of the relaxation being due to the whole NPs. On itself, T₁ reduction can be attributed also to ions,³⁸ but low r₂/r₁ is typical of whole NPs. Compared to the literature,³⁷ it is important to note that we used PLGA with low molecular weight: it is commonplace for the MW of a polymer coating to regulate its mobility and entanglement, and thus the access to its surface.⁵⁴

Table 3. R₁ and R₂ values of MnCl₂ solutions in PBS (pH 7.4, 100 mM) and plasma (4 h of incubation) measured at 37 °C and in a Time-Domain NMR spectrometer working at clinical MRI field (1.5 T). Error always below 1%. From the values, r was also calculated and reported in the table.

conc Mn ²⁺ (mM)	R ₁ in PBS (ms)	R ₁ in plasma (ms)	R ₂ in PBS (ms)	R ₂ in plasma (ms)
0.25	1.5	10.4	19.7	23.1
1	5.5	20.7	66.3	56.9
10	50.9	105.2	609.8	530.2
r (mMol ⁻¹ s ⁻¹)	5.06	9.58	60.47	52.26

Phantom analysis of contrast enhancement using MnO@PLGA NPs by MRI. Longitudinal relaxation times of polymeric NPs were measured by an MR scanner at 7 T to preliminarily investigate the MRI positive contrast power. The T_1 -weighted images showed a clear dose-dependent signal in both PBS and plasma (Fig. 6A,B), in agreement with time-domain NMR (TD-NMR) data. The r_1 of NPs remained constant in PBS, while it increased over time in plasma. MRI analysis thus confirmed TD-NMR data. No contrast was detected in both physiological media and in plasma for control PLGA NPs (data not shown).

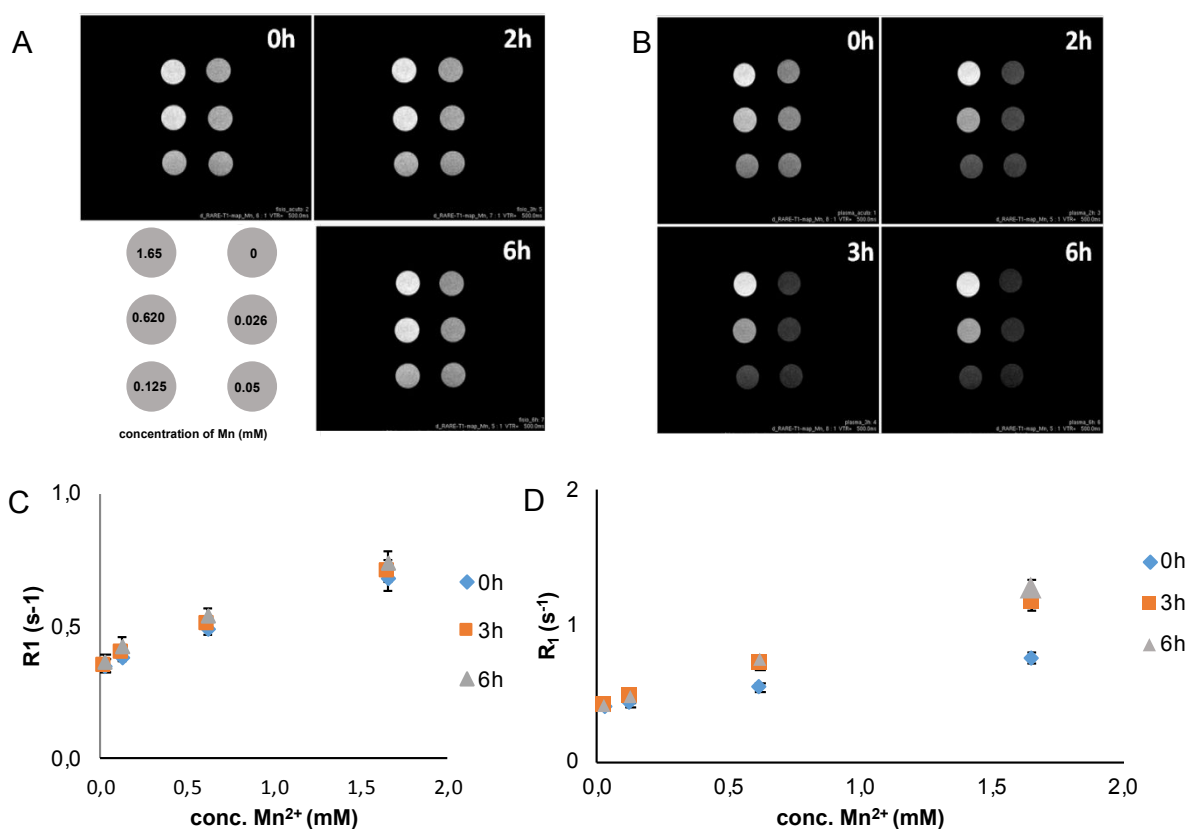


Figure 6. T_1 -weighted images of MnO@PLGA NPs at different concentrations of Mn^{2+} ions (in the range 0-1.65 mM) in (A) PBS and (B) plasma; R_1 of MnO@PLGA NPs in (C) PBS and (D) plasma as a function of Mn^{2+} concentration (0-1.65 mM) and time (0-6 h).

1
2
3 Relaxation rates as a function of Mn^{2+} ion concentration (Fig. 6C,D and S4) clearly demonstrated
4 that the values of r_1 remained stable in PBS for at least 6 h ($0.201 \pm 0.008 \text{ mM}^{-1} \text{ s}^{-1}$). As expected
5 from TD-NMR and MRI studies, the relaxivity of MnO@PLGA NPs increased over time at all
6 tested concentrations, in the presence of plasma (r_1 raised from 0.217 to $0.532 \text{ mM}^{-1} \text{ s}^{-1}$). In
7 particular, it was observed that the initial r_1 in plasma matched the r_1 values in PBS. This
8 observation confirmed that no sample degradation occurred within minutes after MnO@PLGA
9 NPs dispersion in plasma. To check whether the effect of entire nanoparticles or Mn^{2+} ions was
10 prevalent either in PBS or plasma, we measured the release of Mn^{2+} ions in both media. As
11 indicated in Fig. S5, the particles were partly dissolved, but only after several hours and never
12 completely. These data support the opsonisation hypothesis, according to which the time-
13 dependent r_1 enhancement was caused by the release of MnO CA from PLGA NPs in plasma, most
14 likely due to matrix swelling rather than to polymer degradation. From MRI data of MnO@PLGA
15 NPs, 0.026 mM Mn^{2+} was the minimum concentration that could provide a detectable signal.
16
17
18
19
20
21
22
23
24
25
26
27
28
29
30
31

32
33 **Investigation of cytotoxicity profile for MnO@PLGA and MnO@PMA NPs.** The toxicity
34 of Mn-based CAs is primarily due to the action of Mn^{2+} as a calcium channel blocker with effects
35 on muscle electrophysiology and contractility. This effect has limited the use of Mn^{2+} salts as
36 paramagnetic CAs for *in vivo* diagnosis.⁵⁵⁵⁶ In addition, Mn^{2+} complexes are relatively unstable *in*
37 *vivo* and tend to dissociate in biological media. Indeed, Mn ions can be displaced by other divalent
38 cations, such as Ca^{2+} , Mg^{2+} , or Zn^{2+} . Hence, some concerns exist about the potential long-term
39 toxicity associated with the use of Mn CAs.⁵⁷⁵⁸⁵⁹ Therefore, to assess the potential of MnO@PMA
40 and MnO@PLGA NPs as CAs, we investigated their toxicity *in vitro*. Cytotoxicity tests using
41 representative HeLa cancer cells demonstrated that both PLGA NPs and MnO@PLGA NPs were
42 safe in the tested concentration range (Fig. 7). The low toxicity of the NPs was confirmed by ROS
43
44
45
46
47
48
49
50
51
52
53
54
55
56
57
58
59
60

assay (Fig. S6): the amount of ROS, detected after 24h, did not induce any toxic effect, as demonstrated by MTT analysis. Table 4 correlates PLGA weight and Mn²⁺ concentration. After 72 h of incubation, the viability of treated cells was comparable to that of the control. In contrast, MnO@PMA exhibited mild time- and dose-dependent cytotoxicity above 15 nM (Fig. 7), suggesting that the cytotoxicity observed using MnO@PMA NPs was prevented by PLGA coating.

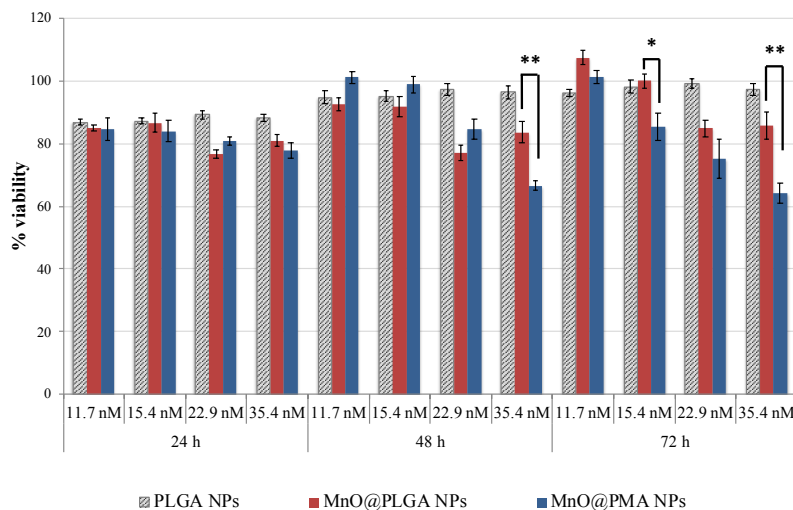


Figure 7. *In vitro* cytotoxicity study in HeLa cells. Cell viability was assessed by MTT assay. Cells treated for 72 h with PLGA NPs; MnO@PLGA NPs; and MnO@PMA NPs. Reported values are the mean of five replicates \pm s.e. normalized on the proliferation of untreated cells. *P <0.05; **P <0.01 (Student's *t*-test).

Table 4. Mn ions concentration relative to nanoparticles dosages used in bioavailability experiments.

[PLGA NPs] (mg mL ⁻¹)	[MnO NPs] (μ g mL ⁻¹)	[Mn(II)] (μ g mL ⁻¹)	[Mn(II)] (nM)
0.05	3.9	0.6	11.7
0.06	4.9	0.8	15.4

0.49	45.4	1.3	22.9
0.98	91.7	1.9	35.4

CONCLUSIONS

In summary, two Mn-based nanoprobcs were investigated as safer and effective positive contrast enhancers, with the objective of providing alternatives to Gd and Teslascan©. MnO@PMA NPs showed higher longitudinal relaxivity compared to Gd-based CAs and Teslascan©,⁶⁰ while maintaining a low r_2/r_1 ratio, both in PBS and plasma, also exhibiting low cytotoxicity. An improved biodegradable Mn-based nanocomposite, consisting of MnO nanoclusters incorporated into PLGA NPs (MnO@PLGA) was synthesized. MnO@PLGA NPs offered the advantage of lower toxicity with 3-fold higher T_1 -contrast power compared to Gd and Teslascan©. Our water-soluble MnO@PLGA CAs exhibited a very pronounced dose- and time-dependent r_1 increase in plasma, possibly due to the use of relatively short PLGA as compared to the literature, thereby favoring the release of active NPs in plasma while maintaining a high particle stability in PBS. Comparison with relaxation phenomena induced by Mn ions in the same solutions, which are characterized by ratios close to 10 and thus very far from the ideal value of 1, indicate that the measured properties are the unique result of the release of entire nanoparticles. This unforeseen property is particularly attractive in view of clinical translation: the localized release of contrast agent at target cells or tissues by either passive (*i.e.*, via the enhanced permeability and retention effect) or active (*i.e.*, exploiting NP functionalization with selective ligands) targeting offers the chance to strongly improve the signal intensity over time, further enhancing MRI sensitivity. The use of these CAs brings new challenges for clinical application. Commercially available CAs are non-specific, whereas PLGA and PMA polymers have been proposed as basic envelopes, suitable

1
2
3 for subsequent surface bioengineering with selective ligands (*e.g.*, small molecules, peptides and
4 antibodies) for molecular recognition associated to specific disorders. In addition, therapeutic
5 agents or secondary imaging components can be loaded into PLGA nanoparticles, resulting in
6 multifunctional theranostics for early detection, diagnosis, and personalized treatments. The
7 combination of all these properties makes MnO@PMA and MnO@PLGA NPs promising
8 nanoprobe for molecular imaging.
9
10
11
12
13
14
15
16
17
18
19
20
21

22 ASSOCIATED CONTENT

23
24 **Supporting Information.** TEM images, DLS size distribution of NPs (Fig. S1), XRD
25 diffractogram of MnO@PMA NPs (Fig. S2), R₁ and R₂ stability of NPs (Fig. S3), release profile
26 of Mn²⁺ (Fig S4) and T₁-weighted images of different concentrations of MnO@PLGA NPs (Fig.
27 S5). This file (PDF) is available free of charge.
28
29
30
31
32
33
34
35
36
37

38 AUTHOR INFORMATION

39 **Corresponding Author**

40
41 * Dr. Miriam Colombo, Università degli Studi di Milano-Bicocca, Dipartimento di Biotecnologie
42 e Bioscienze, Piazza della Scienza, 2 - 20126 Milan, Italy. Phone: +39 02 6448 3388; Email:
43 miriam.colombo@unimib.it
44
45
46
47
48
49

50 **Author Contributions**

51
52 The manuscript was written through contributions of all authors. All authors have given approval
53 to the final version of the manuscript. †These authors contributed equally to this work.
54
55
56
57
58
59
60

Funding Sources

Financial support from the Italian Ministry of University and Research (MIUR) through grant “Dipartimenti di Eccellenza-2017” to University of Milano Bicocca, Department of Biotechnology and Biosciences, and Spanish Ministerio de Ciencia e Innovación (Grant No. MAT2017-86659-R) are acknowledged.

Notes

The authors declare that they have no competing interests.

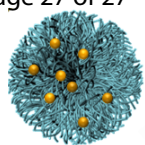
ACKNOWLEDGMENT

We thank T. Canu and A. Esposito (IRCCS San Raffaele Hospital, Milano) for Magnetic Resonance service. We thank Silvia Ferrario (Graftonica) for performing XRD.

ABBREVIATIONS

CA, contrast agent; NP, nanoparticle; MRI, magnetic resonance imaging; PLGA, poly(lactic-co-glycolic) acid; r_1 , longitudinal relaxivity; r_2 , transverse relaxivity; T_1 , spin-lattice relaxation time; T_2 , spin-spin relaxation time; R_1 , longitudinal relaxation rate; R_2 , transverse relaxation rate; PMA, poly[isobutene-*alt*-maleic anhydride] graft dodecylamine (75%) polymer; ICP-OES, Inductively Coupled Plasma Optical Emission Spectrometry; TD NMR, time-domain nuclear magnetic resonance; TEM, transmission electron microscopy; DLS, dynamic light scattering; PBS, phosphate buffer saline; PDI, polydispersity index; ξ , zeta-potential;

REFERENCES



MnO@PLGA

