



PART II: The legacy of isotropic ice

1. A flow law for ice: experimental evidence

2. A flow law for ice: continuum mechanical modeling

3. Microscale processes beyond secondary creep





Experimental in-situ validation of any flow law for ice is difficult:

- extremely long time scales
- Flow-Structure-Environment-Interplay

Experimental validation in the lab suffers from

- limitations in test-duration
- constrained dimensions
- limited/no account of FSEI

BUT:

analogies to high temperature alloys/ metal crystal plasticity (since early 20th century, von Mises etc.)







Ice in the lab: since ~1950Early experiments:



Simple shear, uniaxial tension/compression at constant force (stress) or constant strain rate and prescribed temperature

- For constant stress experiments
 - Applied force is monitored
 - Deformation $\gamma(t)$, $\delta(t)$ is measured
- For constant strain rate experiments
 - Certain components of deformation are monitored
 - Forces are measured

























1. A flow law for ice: experimental evidence Measured creep curves



- a) Creep curves for isotropic polycrystalline ice at stresses between 0.15 MPa and 0.85 MP (Glen, 1953)
- b) Creep curves for isotropic polycrystalline ice at various temperatures and a pressure of 0.6 MPa (Glen, 1953)
- c) Creep curves for isotropic polycrystalline ice at -4.8 deg C at stresses between 0.7 MPa and 1.54 MPa (Steinemann, 1958)





















strain ε



1. A flow law for ice: experimental evidence



Ice can not be regarded as a material with a single unique constitutive response. The latter is determined by the FSEI and is seldomly one-dimensional. Rational Mechanics of Material Modeling provides the framework for general, thermomechanically coupled three-dimensional constitutive modeling of ice.

time t





The behavior of any material body on Earth is such that it obeys

- conservation of mass
- conservation of linear momentum
- conservation of energy

 $\dot{\rho} + \rho \operatorname{div} \mathbf{v} = 0$ $\rho \dot{\mathbf{v}} = \operatorname{div} \mathbf{T} + \rho \mathbf{g}$ $\rho \dot{\varepsilon} = -\operatorname{div} \mathbf{q} + \operatorname{tr}(\mathbf{D} \cdot \mathbf{T}) + \rho r$

Material specific behavior enters through

- restrictions on the spatio-temporal variation of the fields involved (e.g. incompressibility, represented by ρ =0)
- prescription of constitutive relations (e.g. a "flow-law" for the stress)





Constitutive equations

- relate heat flux \boldsymbol{q} and internal energy $\boldsymbol{\epsilon}$ to temperature \mathcal{T}

 $\mathbf{q} = -\kappa(T) \operatorname{grad} T$

$$\dot{\varepsilon}(T) = c_p(T)\dot{T}$$

• relate stress **T** to e.g. velocity \mathbf{v} and temperature \mathcal{T}

$$\mathbf{T} = \hat{\mathbf{T}}(\rho, \mathbf{v}, T)$$

$$\mathbf{T}^{E} = \hat{\mathbf{T}}^{E}(\mathbf{D} = \operatorname{sym}\operatorname{grad} \mathbf{v}, T)$$

$$\mathbf{v} = (v_{x}(x, y, z, t), v_{y}(x, y, z, t), v_{z}(x, y, z, t))$$

$$\mathbf{T} = -p\mathbf{I} + \mathbf{T}^{E} = -p\mathbf{I} + \begin{pmatrix} \tau_{xx} & \tau_{xy} & \tau_{xz} \\ \tau_{xy} & \tau_{yy} & \tau_{yz} \\ \tau_{xz} & \tau_{yz} & \tau_{zz} \end{pmatrix}$$

$$\mathbf{D} = \hat{\mathbf{D}}(\mathbf{T}^{E}, T)$$





Generalizing $\varepsilon = A(T) f(|\sigma|) sgn(\sigma)$ (Glen-Steinemann) to a 3-d law?

Material Modeling: the "Rational Mechancis" approach (C. Truesdell, R.A. Toupin, W. Noll, 1960ies, Hutter & Jöhnk, 2004)







Generalizing $\varepsilon = A(T) f(|\sigma|) sgn(\sigma)$ (Glen-Steinemann) to a 3-d law?

Material Modeling: the "Rational Mechancis" approach (C. Truesdell, R.A. Toupin, W. Noll, 1960ies)

	χ e^3 e^2 e^2	 χ motion function X,x position of particle in reference/present configuration
el	eı	configuration

define a set S of independent, primary variables:

S={ρ, *T*, **v**,...}

 $C = \{ T, q, \varepsilon, ... \}$

- define a set C of dependent, constitutive quantities:
- constitutive relation: $T(X, t) = \check{T}_{Y \text{ in }_{\mathcal{B}}, 0 \leq s < \infty} (\chi(Y, t-s), \rho(Y, t-s), T(Y, t-s), X)$ respects: determinism, inhomogeneity, non-local effects, rule of equipresence Swedish Polar Research Secretariat / Bert Bolin Center for Climate Research / Stockholm University





Simplification of general constitutive relations achieved by application of ...







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$$\mathbf{T}(\mathbf{X}, t) = \mathbf{\check{T}}_{\mathsf{Yin}_{\mathcal{B}}, 0 \le s < \infty} (\chi (\mathsf{Y}, t-s), \rho(\mathsf{Y}, t-s), T(\mathsf{Y}, t-s), \mathsf{X})$$

- "common sense":
- local action

- homogeneity
- fading memory

- ... Rational Mechanics principles:
- invariance under change of observer and rule of material indifference (principle of objectivity)
- principle of material symmetry
- entropy principle (2nd law of thermodynamics)





Example: the stress tensor for a linear elastic solid

functional dependence:



 $\mathbf{F} = \mathbf{F}_{iA} = \partial \mathbf{x}_i / \partial \mathbf{X}_A$





Example: the stress tensor for a linear elastic solid

functional dependence:

material objectivity:

$$T = \check{T}(F)$$
$$T = \check{T}(F) = \check{T}(E)$$

$$\mathbf{F} = \mathbf{F}_{iA} = \partial \mathbf{x}_i / \partial \mathbf{X}_A$$

2E= (F^TF-1) Green-Lagrange strain tensor





Example: the stress tensor for a linear elastic solid

functional dependence:

material objectivity:

entropy principle:

$$T = \check{T}(F)$$
$$T = \check{T}(F) = \check{T}(E)$$
$$S = \rho \,\partial\Psi / \,\partial E$$

$$\mathbf{F} = \mathbf{F}_{iA} = \partial \mathbf{x}_i / \partial \mathbf{X}_A$$

2E= (F^TF-1) Green-Lagrange strain tensor

Ψ Helmholtz energyS Piola-Kirchhoff stress tensor





Example: the stress tensor for a linear elastic solid

functional dependence:

material objectivity:

entropy principle:

linear behavior:

T=Ť(F)
$T=\check{T}(F)=\check{T}(E)$
S = ρ∂Ψ / ∂ E
2 Ψ= Ε (C ⁽⁴⁾ Ε)

$$\mathbf{F} = \mathbf{F}_{_{iA}} = \partial \mathbf{x}_{_i} / \partial \mathbf{X}_{_A}$$

2E= (F^TF-1) Green-Lagrange strain tensor

Ψ Helmholtz energyS Piola-Kirchhoff stress tensor

 Ψ: quadratic in E
 C^{(4):} elasticity tensor with 81 components





Example: the stress tensor for a linear elastic solid (ctnd)

symmetry properties of **C**⁽⁴⁾ (81 independent components):

- **S** and **E** are symmetric tensors: $C^{(4)}$ has 36 independent components
- **S** derivable from the potential Ψ : $\mathbf{C}^{(4)}$ has 21 independent components
- Voigt notation: $\Sigma = C^{*(2)} \Xi$

$$\boldsymbol{\Sigma} = (S_{11}, S_{22}, S_{33}, S_{12}, S_{13}, S_{23}), \qquad \boldsymbol{\Xi} = (E_{11}, E_{22}, E_{33}, E_{12}, E_{13}, E_{23})$$

C^{*(2)} symmetric 6 x 6 matrix





In 3D, a linear elastic solid has at most 21 independent elasticity constants. To reduce this number, the symmetry of the material itself can be exploited.



orthotropic, 9 coefficients Symmetry wrt 180° rotations

c8 0

c9



regular cubic, 3 coefficients Symmetry wrt 90° rotations about 3 fixed perpendicular axes



orthotropic, horizontally regular, 6 coefficients, Symmetry wrt 180° rotations and 90° rotations about a given axis of symmetry



isotropic, 2 coefficients Symmetry wrt arbitrary rotations c1= λ +2 μ , c2= λ



orthotropic, horizontally isotrop ("transverse isotropic"), 5 coefficients Symmetry wrt 180°rotations and arbitrary rotations about a given axis



hexagonal symmetry, 5 coefficients Symmetry wrt n x 30° rotations about given axis (c-axis)





Nye's generalization of Glen's flow law:

Postulate: Cold ice is a density preserving, viscous, heat conducting fluid

Constitutive relation: $\mathbf{T}^{E} = \check{\mathbf{T}} (\mathbf{D}, T, \text{ grad } T) \iff \mathbf{D} = \check{\mathbf{D}} (\mathbf{T}^{E}, T, \text{ grad } T)$

Assumption 1: The dependence on grad *T* has never been measured and is hence dropped. Rational Mechanics Modeling gives

$$D = \beta_1 \mathbf{1} + \beta_2 \mathbf{T}^{E} + \beta_3 \mathbf{T}^{E^2} \qquad \beta_1, \beta_2, \beta_3 = \text{fct}(\text{tr } \mathbf{T}^{E}, \text{tr } \mathbf{T}^{E^2}, \text{det } \mathbf{T}^{E}, T)$$

Incompressibility: div $\mathbf{v} = 0$ or tr $D = 0$
$$0 = 3\beta_1 + \beta_3 \text{tr } (\mathbf{T}^{E^2})$$

$$D = -\beta_3 \text{tr } (\mathbf{T}^{E^2})/3 + \beta_2 \mathbf{T}^{E} + \beta_3 \mathbf{T}^{E^2}$$





Nye's generalization of Glen's flow law (cntd):

Assumption 2: **D** and **T**^E are collinear to each other ($\beta_3 = 0$).

$$\mathbf{D} = \beta_2 \mathbf{T}^{\mathbf{E}} \qquad \beta_2 = \text{fct}(\text{tr } \mathbf{T}^{\mathbf{E}} = 0, \text{ tr } \mathbf{T}^{\mathbf{E}^2}, \text{ det } \mathbf{T}^{\mathbf{E}}, T)$$

Assumption 3: D does not depend on det T^E

$$\mathbf{D} = \beta_2 \mathbf{T}^{\mathsf{E}} = \beta_2 (\text{tr } \mathbf{T}^{\mathsf{E}^2}, T) \mathbf{T}^{\mathsf{E}}$$

Assumption 4: β_2 can be factorized as β_2 (tr $\mathbf{T}^{\mathbf{E}^2}$, T) = A(T) f (tr $\mathbf{T}^{\mathbf{E}^2}$)

$$\mathbf{D} = \mathbf{A}(T) f (\text{tr } \mathbf{T}^{\mathbf{E}^{2}}) \mathbf{T}^{\mathbf{E}}$$

3D generalization of the flow law of Glen & Steinemann (Nye 1952)







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strain ε











Properties of H₂0

Water has the greatest diversity (polymorphism) of solid phases over all known substances:

- 2 amorphous phases (lacking crystalline structure)
- 16 crystalline phases (ice lh, lc, II - XV)

Ice Ih possesses hexagonal symmetry

• atomic packing factor: < 34%

2500УII VIII 2000 ^oressura, MPa 15001 \mathcal{M} 1006 Liquid. 50a -100 -50 th. 5Q., 00 Temperature, °C





Microstructure of ice Ih

- hexagonal symmetry
- concentration of molecules in *basal planes*
- weak O=O bonding in between basal planes ("hard glide" along pyramidal/prismatic planes
- strong O=O bonding within the basal planes ("easy glide" along basal planes)
- c-axis: perpendicular to basal planes







3. Microscale processes beyond secondary creep Microstructure of ice Ih

strong O=O bonding within the basal planes ("easy glide" along basal planes)

- slip resistance along basal planes is up to 60 times smaller than in other slip systems (210 K < T < 273.15 K)
- the "deck of cards" analogy dates back to McConell (1891)

the crystal behaved as if it consisted of an infinite number of indefinitely thin sheets of paper, normal to the optic axis, attached to each other by some viscous substance which allowed one to slide over the next with great difficulty. This comparison proved to be the key to the whole question of the plasticity of a crystal of ice.

 slip along basal planes takes place via dislocation glide and is regarded as the dominant deformation mechanism





Dislocations and other imperfections: Hypothetical vs. real crystals

- Distribution of hydrogens in the oxygen lattice:
 2 hydrogen nuclei close to any oxygen but only one per joining line (Bernal-Fowler rule)
 Violation of these rules: point defects (Bjerrum)
- Discontinuity/offset in the crystal structure: line (1d) / plane (2d) /gross (3d) defects

line defects (1d): dislocations

dislocation density ρ_d (total length of line defect per unit volume [l⁻²]) evolves in time and can lead to strain softening/hardening behavior of ice















measuring c-axis rotation

thin sections under crossed polarizers

- crystals rotate incoming polarized light depending on c-axis orientation
- differently oriented crystals appear with different colors
- orientations are plotted in Schmidt diagrams











Classification of microscale deformation mechanisms

lattice mechanisms: relate to the behavior of the crystalline structure (c-axis evolution)

"fabric"

boundary mechanisms: act at the grain boundaries and relate to grain size (Grain boundary migration, nucleation, recovery, polygonization,)

"texture"

crystal-size dependent isotropic rheology: Barnes (1972), Goldsby & Kohlstedt (1997)





Homogenization of microscopic deformation within a RVE

Consideration of basal glide alone during deformation:

How to deal with geometric misfits occurring between neighboring crystals?

- Taylor's hypothesis (1938, "Plastic strain in metals"): all crystals in an RVE suffer the same strain now discarded for polycrystalline ice
- Sachs' hypothesis (1928, "Zur Ableitung einer Fliessbedingung" [On the derivation of a yield criterion]): all crystals in the RVE suffer the same stress
- VPSC (ViscoPlastic Self Contained; Castelnau, Duval, Lipenkov,... ~1996): combining Taylor and Sachs: stress equilibrium and strain compatibility are recursively satisfied difficulties when formulating a macroscopic flow law not extendable to non-liner stress-deformation relations