Nanoscale

PAPER

Cite this: Nanoscale, 2013, 5, 4355

Received 11th March 2013 Accepted 17th March 2013

DOI: 10.1039/c3nr01244a

www.rsc.org/nanoscale

Introduction

Controlling the morphology of noble metal nanoparticles, gold and silver in particular, has been the subject of intense research interest over the past several decades, owing to the strong size and shape dependence of their properties,¹⁻⁵ leading to exciting applications in a wide variety of fields such as plasmonics,^{1,6-8} catalysis,⁹ electronics¹⁰ and biosensing.¹¹ Various synthetic methods have been developed to prepare noble metal nanoparticles in a wide range of sizes and morphologies in hydrophilic (polar) solvents^{2,4,5,12,13} such as water, DMF and ethylene glycol. It can be safely stated that one can synthesize Au and Ag nanoparticles of any required morphology and dimensions by using existing sets of rules for colloidal synthesis in aqueous

^aBioNanoPlasmonics Laboratory, CIC biomaGUNE, Paseo de Miramón 182, 20009 Donostia - San Sebastián, Spain. E-mail: lpolavarapu@cicbiomagune.es

Growth and galvanic replacement of silver nanocubes in organic mediat

Lakshminarayana Polavarapu*a and Luis M. Liz-Marzán*abc

Although metal nanoparticles with various shapes can be prepared in polar organic solvents, little has been advanced toward the shape-controlled synthesis in non-polar solvents. We report a simple method for the synthesis of nearly monodisperse single crystalline silver nanocubes in a non-polar solvent (1,2-dichlorobenzene) by using oleylamine as both a reducing and capping agent. Mechanistic studies based on the time evolution of Ag nanoparticles revealed that multiply twinned nanocrystals form at the beginning of the reaction, which are gradually transformed into single crystalline Ag nanocubes by oxidative etching. Control experiments showed that the solvent plays an important role in the formation of such single crystalline Ag nanocubes. The effects of reaction temperature, oleylamine concentration, solvent, and the nature of the silver ion precursor on the morphology and monodispersity of the nanoparticles were systematically investigated. Additionally, the galvanic replacement reaction with HAuCl₄ in an organic medium was implemented to prepare hydrophobic hollow Au–Ag nanocages with tunable localized surface plasmon resonances.

solution.^{2,4,14} However, the size and shape controlled synthesis of Au and Ag nanoparticles with tunable optical properties in organic media is still challenging because the reduction of metal ions in organic solvents typically yields multiply twinned spherical nanoparticles due to fast homogeneous nucleation.15-19 In addition, most of the conventional precursors used for nanoparticle synthesis are inorganic salts, which are insoluble in organic solvents.16 Surfactants (containing amines or thiols) are thus required to pull such ions into organic solution by forming organometallic complexes. However, the preparation of metal nanoparticles in organic media usually has several advantages that include narrow size distribution,15,18 large scale production,16 surface functionalization with a variety of organic capping agents,¹⁸ catalysis for organic reactions,^{20,21} self assembly at the interface3 and potential for the fabrication of flexible plasmonic substrates.22

In spite of its difficulty, some examples exist on the synthesis of single-crystalline ultrathin Au nanowires,^{8,23-26} Au–Ag bime-tallic nanowires,²⁷ Au nanorods²⁸ and various shapes of Pd nanoparticles²⁹ in organic solution. In the particular case of Ag, there is a strong tendency to form twinned seed particles during the early stages of the reaction, both in hydrophilic and in hydrophobic solvents.^{5,19} It is thus difficult to prepare single crystalline Ag nanostructures with well defined-shapes. To overcome this problem, Xia *et al.* reported the polyol reduction method for the formation of single crystalline Ag nanocubes by oxidative etching of twin defects in a hydrophilic solvent (the polyol) using chloride containing salts such as HCl or NaCl.⁵ However, such oxidative etchants are immiscible with hydrophobic solvents, thereby limiting the synthesis of Ag nanocubes

RSCPublishing

^bIkerbasque, Basque Foundation for Science, 48011 Bilbao, Spain

^cDepartamento de Química Física, Universidade de Vigo, 36310 Vigo, Spain. E-mail: llizmarzan@cicbiomagune.es

[†] Electronic supplementary information (ESI) available. TEM and HRTEM of Ag nanocubes obtained with 0.44 mol DCB; HRTEM of Ag NPs formed after 1 h of reaction; fraction of twinned vs. single crystalline Ag NPs at different reaction times; TEM images of Ag NPs obtained under an inert atmosphere; TEM image of Ag NPs obtained with 3 mmol and 12 mmol oleylamine; extinction spectra and TEM images of Ag NPs obtained in different solvents; extinction spectra and TEM images of Ag NPs obtained in different amounts of DCB; HRTEM image of a Ag nanocube; extinction spectra and TEM images of Ag NPs obtained using CF₃COOAg and CH₃COOAg as precursors; EDS spectra of a single Au–Ag nanocage; TEM images of fragmented Au NPs; extinction spectra of Ag nanocube dispersions after galvanic replacement at 100 °C; extinction spectra and TEM images of Au–Ag NPs obtained after galvanic replacement at room temperature. See DOI: 10.1039/c3nr01244a

in non-polar organic media. To the best of our knowledge, there are only two reports in the literature for the synthesis of Ag nanocubes in organic media.^{30,31} However, these methods are rather sensitive to the concentration of oxidative etchants and often result in a mixture of spherical and cubic Ag NPs. Peng and Sun³¹ reported the synthesis of Ag nanocubes in a binary hydrophobic solvent by using an amphiphilic oxidative etching agent, dimethyldistearylammonium chloride (DDAC), in the reaction medium. Metal salts such as FeCl₃ and Fe(acac)₃ have also been used as oxidative etching agents for the synthesis of 13 nm single crystalline Ag nanocubes in isoamyl ether, as reported by Xia *et al.*³⁰ In the above two methods, ether was chosen as the solvent to dissolve the inorganic salts as they can interact with the oxygen atom in ether.

In this work, we demonstrate a simple and optimised method for the synthesis of single crystalline Ag nanocubes in an organic solvent (1,2-dichlorobenzene (DCB)) by using oleylamine as both a reducing and capping agent, with no need for additional oxidative etching inorganic salts. We found that the compositions of the solvent and Ag precursor, as well as temperature, play important roles on the final morphology of the obtained Ag nanoparticles. Additionally, we demonstrate that Ag nanocubes can be transformed into Au–Ag nanocages *via* a galvanic replacement reaction with HAuCl₄, even in organic solvents, leading to accurate control of the LSPR position.

Experimental section

Chemicals

Silver nitrate (AgNO₃), silver acetate (CH₃COOAg), silver trifluoro acetate (CF₃COOAg), HAuCl₄·3H₂O, oleylamine (OA) (technical grade, 70%), chlorobenzene, 1-octadecene and toluene were purchased from Sigma-Aldrich. 1,2-Dichloro benzene (DCB; 99% extra pure) was purchased from Acros Organics. All chemicals were used as received. **Caution**: DCB is highly toxic and should be handled with care in an appropriate fume hood.

Synthesis of silver nanocubes

0.6 mmol of AgNO3 and 6 mmol of oleylamine were dissolved in 50 mL (0.44 mol) of DCB and the reaction mixture was sonicated until AgNO₃ was completely dissolved. The solution colour changed from colourless to pale yellow upon sonication. The resulting solution was transferred into an oil bath and the bath temperature was slowly raised up to 165 °C at a rate of 5 °C min⁻¹ and then the reaction solution gradually acquired a deep yellow colour. The reaction mixture was kept for 48 hours at 165 °C with continuous stirring under atmospheric conditions and then slowly cooled down to room temperature. The resulting solution contained nearly monodisperse single crystalline silver nanocubes. To study the time evolution of Ag nanoparticles during the reaction, small aliquots were taken from the reaction mixture at different reaction times by using a Pasteur pipette and analysed by UV-visible spectroscopy and transmission electron microscopy (TEM). Control experiments were carried out using the same procedure to investigate the

It was found that the reaction time and yield of Ag nanocubes strongly depends on the reaction volume. When the reaction was carried out in 100 mL (0.88 mol) of DCB, while keeping the amount of oleylamine and $AgNO_3$ constant, the reaction was completed in 12 h and the monodispersity and yield of Ag nanocubes were increased. However, when the reaction was carried out in 25 mL (0.22 mol) of DCB, the product mainly contained twinned Ag nanoparticles with a small percentage of Ag nanocubes, even after 48 h of reaction time.

Galvanic replacement reaction

In a typical synthesis, 1 mL of the as prepared (using 0.6 mmol AgNO₃, 6 mmol oleylamine and 0.88 mol DCB) silver nanocube solution was injected into a round-bottom flask containing 19 mL of toluene under magnetic stirring and the system was heated to 100 °C in an oil bath. A fresh HAuCl₄ stock solution was prepared by dissolving 9 mg of HAuCl₄·3H₂O in 6 mL of toluene and 0.4 mL of oleylamine and stored in the dark before use. Different volumes (0.1, 0.15, 0.2, 0.25, 0.3, 0.35, 0.4 and 0.8 mL) of the HAuCl₄ stock solution were added to the boiling reaction mixture under magnetic stirring, so that the final Au : Ag molar ratios were ~6%, 9%, 12%, 15%, 18%, 21%, 24% and 50%, respectively. After each addition of HAuCl₄ the stirring was continued for another 10 min until the UV-Vis spectra did not change further. Galvanic replacement was also carried out at room temperature to study the effect of reaction temperature, while keeping the other parameters unchanged. The obtained particles were purified by centrifugation and redispersed in toluene for TEM characterization.

Characterization

The Ag nanoparticles were characterized by UV-visible spectroscopy (Agilent 8543, 1 cm path length quartz cuvettes) and TEM (using a JEOL JEM 2010F, field emission electron microscope, operating at 200 kV). Energy-dispersive X-ray spectroscopy (EDS) spectra were obtained by using an INCA X-Sight spectrometer (Oxford instruments), within the TEM instrument. For TEM analysis, the obtained Ag nanocubes were first purified by centrifugation and re-suspended in toluene. Then, a drop of the purified nanoparticle solution was spotted onto a carbon coated copper grid and dried in a fume hood at room temperature.

Results and discussion

Growth of Ag nanocubes

The synthesis of hydrophobic Ag nanocubes is based on the reduction of $AgNO_3$ with oleylamine at high temperature in DCB solution (see Experimental section for details). Fig. 1e shows a



Fig. 1 TEM images of the Ag nanoparticles obtained at (a) 1, (b) 4, (c) 8, (d) 24 and (e) 48 hours of reaction time in 0.44 mol of DCB solution (0.6 mmol of AgNO₃ and 6 mmol of oleylamine) and their corresponding extinction spectra (f). HRTEM image of the Ag nanoparticles formed after 1 h of reaction time is shown in Fig.S2.†

typical TEM image of Ag nanocubes obtained after 48 h of reaction in 0.44 mol of DCB solution (0.6 mmol of AgNO₃ and 6 mmol of oleylamine), clearly showing nearly monodisperse Ag nanocubes with slight truncation and 26 ± 2 nm average edge length (standard deviation 1.6%). Additional TEM images obtained under lower magnification confirm the monodispersity, while HRTEM shows that they are single crystalline (see Fig. S1†). The slight truncation of the Ag nanocubes arises due to partial oxidation and release of Ag atoms at the corners during the synthesis, but the cubic shape is preserved upon storage in non-polar solvents for extended periods of time (at least 10 months). This is in contrast with the tendency of Ag nanocubes prepared by polyol reduction, which tend to become increasingly rounded when they are stored in water for shorter periods of time.³²

To investigate the growth mechanism responsible for the formation of single crystalline Ag nanocubes, we monitored their growth by extracting aliquots from the reaction mixture at different reaction times and analysing them by UV-visible spectroscopy and TEM (after diluting 10 μ L aliquots in 2 mL of

toluene). Fig. 1 shows representative TEM images of Ag nanoparticles obtained at different reaction times, as well as their corresponding extinction spectra. As shown in Fig. 1a, the Ag nanoparticles formed after one hour of reaction are spherical, with sizes ranging from 8 to 10 nm and with the corresponding extinction maximum at 414 nm. High resolution TEM images (Fig. S2⁺) revealed a multiply twinned structure for these nanoparticles. As the reaction progresses to t = 4 h, multiply twinned quasi spherical Ag nanoparticles with a wider size distribution (10-20 nm) coexist with a small population of single crystalline polyhedral Ag nanoparticles, including nanocubes, as observed in the TEM images (Fig. 1b). During this initial step, the extinction spectrum broadened due to a broader particle size distribution and the LSPR maximum redshifted to 416 nm, which is attributed to an increase in both particle size and anisotropy. As the reaction proceeds further, the population of twinned particles decreases whereas the population of single crystalline polyhedral nanoparticles and nanocubes gradually increases, as reflected in the TEM images (Fig. 1c-d). The sample obtained after 24 h (Fig. 1d) essentially consists of single crystalline Ag nanocubes with a small percentage of multiply twinned particles and the corresponding extinction spectrum exhibits a red-shifted sharp LSPR centred at 422 nm, corresponding to Ag nanocubes of \sim 26 nm lateral size. These results suggest that the size of the initially obtained multiply twinned Ag nanoparticles increases by continuous reduction of Ag^+ ions by oleylamine, but apparently the growth stops after reaching a critical size and then the nanoparticles start to transform into single crystalline nanoparticles by oxidative etching of defect sites. As shown in Fig. 1d, the size of the twinned particles formed after 24 h is the same as that of the single crystal nanocubes. When the reaction time was further extended to 48 h, all the twinned particles were completely converted into single crystal nanocubes (a histogram of the estimated population distribution of twinned vs. single crystalline NPs as a function of reaction time is shown in Fig. S3⁺) and thus the final product essentially contains truncated nanocubes of ~26 nm average edge length, with an LSPR of 422 nm and a shoulder at 362 nm (Fig. 1e and f), which are attributed to a dipolar plasmon resonance and a quadrupolar mode, respectively.33-35 Regarding the observed crystallinity change from multiply twinned nanoparticles to single crystals, it has been previously reported for the synthesis of Ag nanocubes in aqueous solutions, in relation with the oxidative etching of highly reactive defect regions by etchants such as HCl or NaCl.5,12 In the present organic phase synthesis, additional oxidative etchants were not added, but NO3⁻ and Cl⁻ ions from the silver salt and the solvent (DCB), as well as dissolved oxygen, could initiate the slow oxidative etching and dissolution of defect sites at a relatively high temperature and prolonged reaction time, thereby leading to the formation of the final single crystalline Ag nanocubes.5 In previous works it has been reported that NO₃⁻ and Cl⁻ ions could initiate the etching of twinned Ag NPs and their transformation into single crystalline NPs in the absence of dissolved oxygen.31,36 We tested this hypothesis by means of a control synthesis in which the reaction mixture was purged by bubbling nitrogen gas and then

reaction was carried out under nitrogen while the other experimental parameters were the same as indicated in Fig. 1. TEM characterization (Fig. S4[†]) showed that most of the obtained NPs after 48 h of reaction are Ag nanocubes, with a small population of twinned NPs and nanorods. These results suggest that etching can indeed occur in the absence of oxygen, but the presence of some twinned Ag NPs indicates the reaction was slower. However, when the reaction was carried out in atmospheric conditions all the twinned Ag NPs were converted into single crystalline Ag nanocubes (Fig. S1[†]).

Effect of reaction parameters

The reducing ability of oleylamine and the oxidative etching capability of NO_3^- and Cl^- ions usually increase with increasing the reaction temperature.³¹ Therefore, we studied the effect of reaction temperature on the morphology of the Ag nanoparticles by running the reaction at different temperatures (120, 140, 165 and 185 °C) while other parameters were the same as those used for the experiment described in Fig. 1. Representative TEM images are shown in Fig. 2, from which we found that the optimum temperature for the synthesis of Ag nanocubes was 165 °C (Fig. 2c). When the temperature was further increased up to 185 °C, the product contained a mixture of nanocubes, nanospheres and few nanorods (Fig. 2d), which might be related to more extensive etching and truncation of the nanocubes.

A similar effect was observed when the amount of oleylamine was decreased to 3 mmol (Fig. S5a[†]). In this case, insufficient protection of the Ag nanocubes surface may lead to further truncation into spherical nanoparticles. However, when more oleylamine was used (12 mmol), polydisperse MT nanoparticles were formed (Fig. S5b[†]), which is likely due to strong protection of the initially formed NPs, preventing the etching of defect sites.

Interestingly, the nature of the solvent was found to play a significant role on the morphology of the obtained nanoparticles. Indeed, when solvents such as toluene, chlorobenzene and octadecene were used (Fig. S6[†]), an increase in the



Fig. 2 TEM images of the Ag nanoparticles synthesized at different reaction temperatures: (a) 120, (b) 140, (c) 165 and (d) 185 °C, while other reaction conditions were kept the same as in Fig. 1 (0.6 mmol of AgNO₃ and 6 mmol of oleylamine in 0.44 mol of DCB solution).



Fig. 3 (a and b) Extinction spectrum and TEM image of nanocubes prepared in 0.88 moles of DCB solution by Ag NO_3 reduction. (c) TEM image of nanocubes prepared in 0.88 moles of DCB solution using CF₃COOAg as a precursor. Other reaction parameters were the same as those in Fig. 1 (0.6 mmol AgNO₃, 6 mmol oleylamine). (d) HRTEM image of a Ag nanocube. The inset is the FFT spot pattern, indicating highly symmetric spots, which can be indexed to (110), (200) and (020) reflections of face-centred cubic silver.

reaction yield was noted for high boiling point solvents (Fig. S6a[†]). However, the Ag nanoparticles obtained in such solvents were polycrystalline and did not display cubic morphology. The main difference between DCB and the other solvents is a higher content of chloride atoms as well as a lower C–Cl bond dissociation energy of DCB as compared to chlorobenzene,³⁷ resulting in a higher tendency to release chloride ions at high temperature, which can eventually etch the twinned Ag nanoparticles.^{5,12}

In addition, the influence of reactants concentration was studied by carrying out the synthesis in different amounts (0.22 and 0.88 moles) of DCB, while keeping the amounts of AgNO₃ and oleylamine the same as in the experiment described in Fig. 1. The extinction spectra and TEM images of the obtained Ag nanoparticles in different amounts of DCB are summarized in Fig. S7.[†] The results indicate that Ag nanocubes with higher quality and yield can be prepared by reducing 0.6 mmol of AgNO3 with 6 mmol of oleylamine in 0.88 moles of DCB (Fig. 3b and S7d[†]). Fig. 3a shows the extinction spectra of the Ag nanocubes obtained in 0.88 moles of DCB, with a dipolar LSPR at 424 nm and a quadrupolar band at 365 nm.³⁴ The TEM image (Fig. 3b) clearly shows uniform contrast over each nanocube, suggesting that they are monocrystalline, which was confirmed by HRTEM on individual particles (Fig. 3d and S8⁺). Finally, the reaction was tested with other silver ion precursors such as CF₃COOAg and CH₃COOAg. Monodisperse Ag nanocubes with 23 nm average edge length were obtained using CF₃COOAg (Fig. 3c), with high monodispersity and yield

(Fig. S9[†]). In contrast, CH₃COOAg produced quasi spherical Ag nanoparticles (Fig. S10[†]).

Galvanic replacement and formation of Au nanocages

Galvanic replacement reactions have been an effective strategy for the preparation of nanostructures, in which two metals are exchanged because of the difference in their electrochemical potentials, so that a hole is generated due to oxidative dissolution of metal nanocrystals, which subsequently act as templates.38-40 This strategy has been widely applied to the reaction (in polar solvents) of Ag nanoparticles of various shapes with HAuCl₄ to synthesize hollow nanoparticles with tuneable optical properties.^{38,40,41} In this reaction, Ag reacts with HAuCl₄ at an atomic ratio of 3 : 1. However, application of this process in organic media has been restricted to spherical Ag nanoparticles, mainly due to the limitations in the synthesis of shape controlled Ag nanoparticles in these solvents.42-44 We evaluated the application of the galvanic replacement reaction to our Ag nanocubes in a non-polar environment, at different reaction temperatures. Shown in Fig. 4 are representative TEM images of the Ag nanocubes before and after titrating with different amounts of HAuCl₄ solution in toluene 0.1, 0.2, 0.25 and 0.35 mL, so that the final Au : Ag molar ratios were 6%, 12%, 15% and 21%, respectively. The reaction was carried out at 100 °C. It is clear from the images that the initial single crystalline Ag nanocubes (edge length ca. 24 nm) readily display signs of galvanic replacement reaction (through a hole on one of their faces) upon addition of a small amount (6 mol%) of $HAuCl_4$ (Fig. 4b), resembling the usual galvanic replacement reactions in aqueous solution.³⁸ For larger amounts of HAuCl₄, the void size gradually increased, while the corners became further truncated, likely because of dealloying, until ultimately the particles were converted into cage like structures (Fig. 4d). From the TEM images, the particles appear to have different morphology, which is due to the orientation of NPs on either (100) or (111) facets, depending on their degree of corner truncation,45 and HRTEM characterization indicates that the nanocages retained the monocrystallinity of the original cubes (Fig. 4; see also Fig. S11[†] for a larger image). The composition of the nanocages was confirmed by EDS on a single particle, to contain Au and Ag, probably as an alloy (Fig. S12⁺). Interestingly, upon further addition of HAuCl₄, well defined voids were observed at the corners, so that most of the particles appeared to display a nanoframe morphology. The average size of the nanoparticles slightly increased up to 28 nm, indicating the deposition of gold atoms on the exterior surfaces as Ag was depleted from the interior, but the monodispersity was preserved (see Fig. S13[†]). The stability of the obtained nanocages in nonpolar solvents, as well as their high surface-tovolume ratio are expected to render them very useful as catalysts for organic reactions. It is important to note however that excess addition of $HAuCl_4$ ([Au] : [Ag] = 50 mol%) leads to the fragmentation of Au nanostructures (Fig. S14⁺). We additionally found that when the galvanic replacement reaction was carried out at room temperature, rather different morphologies were obtained, where small protuberances formed from the initial stages, finally leading to irregular Au nanoframes (Fig. S15⁺). We speculate that the reason behind this different reshaping process is the decrease of interdiffusion rate of Au and Ag atoms at room temperature.

The morphological and composition changes involved in the galvanic replacement reaction lead to changes in the LSPR of the particles, so that the reaction could be monitored by UV-visible spectroscopy. Fig. 5 shows a photograph of the dispersions of Ag nanoparticles in toluene, before and after galvanic replacement



Fig. 4 Synthesis of Au–Ag hollow nanostructures in an organic medium by the galvanic replacement reaction between Ag nanocubes and HAuCl₄. TEM images of the Ag nanocubes before (a) and after titration with different volumes of HAuCl₄ solution: 0.1 (b), 0.2 (c), 0.25 (d) and 0.35 (e) mL, with final Au : Ag molar ratios of \sim 6%, 12%, 15%, and 21%, respectively. (f) HRTEM image of one Au–Ag nanocage from (d), see Fig. S11 for a large size image.†



Fig. 5 Photograph and normalized extinction spectra of nanocube dispersions before (I) and after (II–VIII) galvanic replacement with different volumes of HAuCl₄: 0.1, 0.15, 0.2, 0.25, 0.3, 0.35 and 0.8 mL, with final Ag : Au molar ratios of ~6%, 9%, 12%, 15%, 18%, 21%, 50%, respectively. The spectra prior to normalization are shown in Fig. S16.†

with different amounts of HAuCl₄, along with their normalized extinction spectra. The colour of the dispersions changes as yellow-orange-red-violet-blue-green-pink with increasing amounts of HAuCl₄ and such colour changes are reflected in the significant LSPR redshifts (Fig. 5; see also the extinction spectra prior to normalization in the ESI, Fig. S16†). However, the extinction spectra are rather narrow, indicating that the obtained particles are quasi-monodisperse, as evidenced by TEM (Fig. 4). Again, excess addition of HAuCl₄ leads to a blue shift due to fragmentation of the hollow nanostructures.

Conclusion

We have developed a simplified method for the preparation of single-crystalline Ag nanocubes in organic solvents by using oleylamine as a reducing and capping agent. The growth mechanism was studied by means of time-dependent TEM and UVvisible spectroscopy characterization. It was found that twinned seed particles are formed at early stages, which eventually transform into single crystalline Ag nanocubes by preferential oxidative etching of twinned particles. Furthermore, galvanic replacement reaction between Ag nanocubes and HAuCl₄ in an organic medium was successfully accomplished, leading to the formation of hollow alloy nanostructures (nanocages and nanoframes) with distinct LSPR peaks, in agreement with similar studies in polar solvents. We expect that the high quality of these hydrophobic Ag nanocubes and AgAu hollow nanostructures will be very useful for a number of applications such as catalysis, photonics, nonlinear optics and sensing.

This work has been funded by the European Research Council (ERC Advanced Grant # 267867, PLASMAQUO). We thank Jorge Pérez-Juste, Isabel Pastoriza-Santos, Akshaya Kumar Samal, Ana Sánchez-Iglesias, and Stefanos Mourdikoudis for helpful discussions. Marco Möller is acknowledged for assistance with HRTEM and EDS characterization.

Notes and references

- 1 L. M. Liz-Marzan, Langmuir, 2006, 22, 32-41.
- 2 M. Grzelczak, J. Perez-Juste, P. Mulvaney and L. M. Liz-Marzan, *Chem. Soc. Rev.*, 2008, **37**, 1783–1791.
- 3 A. Sanchez-Iglesias, M. Grzelczak, J. Perez-Juste and L. M. Liz-Marzan, Angew. Chem., Int. Ed., 2010, 49, 9985– 9989.
- 4 Y. G. Sun and Y. N. Xia, Science, 2002, 298, 2176-2179.
- 5 B. Wiley, T. Herricks, Y. G. Sun and Y. N. Xia, *Nano Lett.*, 2004, 4, 1733–1739.
- 6 S. P. Zhang, K. Bao, N. J. Halas, H. X. Xu and P. Nordlander, *Nano Lett.*, 2011, **11**, 1657–1663.
- 7 W. H. Wang, Q. Yang, F. R. Fan, H. X. Xu and Z. L. Wang, *Nano Lett.*, 2011, **11**, 1603–1608.
- 8 H. J. Feng, Y. M. Yang, Y. M. You, G. P. Li, J. Guo, T. Yu, Z. X. Shen, T. Wu and B. G. Xing, *Chem. Commun.*, 2009, 1984–1986.
- 9 P. Christopher, H. L. Xin and S. Linic, *Nat. Chem.*, 2011, 3, 467–472.
- 10 S. De, T. M. Higgins, P. E. Lyons, E. M. Doherty, P. N. Nirmalraj, W. J. Blau, J. J. Boland and J. N. Coleman, ACS Nano, 2009, 3, 1767–1774.
- 11 B. Sepulveda, P. C. Angelome, L. M. Lechuga and L. M. Liz-Marzan, *Nano Today*, 2009, 4, 244–251.
- 12 S. H. Im, Y. T. Lee, B. Wiley and Y. N. Xia, *Angew. Chem., Int. Ed.*, 2005, 44, 2154–2157.
- 13 I. Pastoriza-Santos and L. M. Liz-Marzan, *Langmuir*, 2002, 18, 2888–2894.
- 14 M. R. Langille, M. L. Personick, J. Zhang and C. A. Mirkin, J. Am. Chem. Soc., 2012, 134, 14542–14554.
- 15 M. Chen, Y. G. Feng, X. Wang, T. C. Li, J. Y. Zhang and D. J. Qian, *Langmuir*, 2007, 23, 5296–5304.
- 16 L. Polavarapu, K. K. Manga, K. Yu, P. K. Ang, H. D. Cao, J. Balapanuru, K. P. Loh and Q. H. Xu, *Nanoscale*, 2011, 3, 2268–2274.
- 17 L. Polavarapu and Q. H. Xu, *Nanotechnology*, 2009, **20**, 185606–185612.
- 18 H. Hiramatsu and F. E. Osterloh, *Chem. Mater.*, 2004, 16, 2509–2511.
- 19 Y. Sun, Chem. Soc. Rev., 2013, 42, 2497-2511.
- 20 H. Lee, S. E. Habas, S. Kweskin, D. Butcher, G. A. Somorjai and P. D. Yang, Angew. Chem., Int. Ed., 2006, 45, 7824–7828.
- 21 X. Hong, D. Wang, S. Cai, H. Rong and Y. Li, J. Am. Chem. Soc., 2012, 134, 18165–18168.
- 22 L. Polavarapu and L. M. Liz-Marzan, *Phys. Chem. Chem. Phys.*, 2013, **15**, 5288–5300.
- 23 Z. Y. Huo, C. K. Tsung, W. Y. Huang, X. F. Zhang and P. D. Yang, *Nano Lett.*, 2008, 8, 2041–2044.

- 24 X. M. Lu, M. S. Yavuz, H. Y. Tuan, B. A. Korgel and Y. N. Xia, *J. Am. Chem. Soc.*, 2008, **130**, 8900–8901.
- 25 N. Pazos-Perez, D. Baranov, S. Irsen, M. Hilgendorff, L. M. Liz-Marzan and M. Giersig, *Langmuir*, 2008, 24, 9855–9860.
- 26 C. Wang, Y. J. Hu, C. M. Lieber and S. H. Sun, *J. Am. Chem. Soc.*, 2008, **130**, 8902–8903.
- 27 X. Hong, D. S. Wang, R. Yu, H. Yan, Y. Sun, L. He,
 Z. Q. Niu, Q. Peng and Y. D. Li, *Chem. Commun.*, 2011,
 47, 5160–5162.
- 28 Z. Q. Li, J. Tao, X. M. Lu, Y. M. Zhu and Y. N. Xia, *Nano Lett.*, 2008, **8**, 3052–3055.
- 29 Z. Q. Niu, Q. Peng, M. Gong, H. P. Rong and Y. D. Li, *Angew. Chem., Int. Ed.*, 2011, **50**, 6315–6319.
- 30 Y. Y. Ma, W. Y. Li, J. Zeng, M. McKiernan, Z. X. Xie and Y. N. Xia, J. Mater. Chem., 2010, 20, 3586–3589.
- 31 S. Peng and Y. G. Sun, Chem. Mater., 2010, 22, 6272-6279.
- 32 Q. A. Zhang, W. Y. Li, L. P. Wen, J. Y. Chen and Y. N. Xia, *Chem.-Eur. J.*, 2010, **16**, 10234–10239.
- 33 A. Bottomley, D. Prezgot, A. Staff and A. Ianoul, *Nanoscale*, 2012, 4, 6374–6382.

- 34 M. A. Mahmoud and M. A. El-Sayed, J. Phys. Chem. C, 2008, 112, 14618–14625.
- 35 Y. H. Lee, H. Chen, Q.-H. Xu and J. Wang, *J. Phys. Chem. C*, 2011, **115**, 7997–8004.
- 36 Y. Tang and M. Ouyang, Nat. Mater., 2007, 6, 754-759.
- 37 X. H. Li, Z. X. Tang and X. Z. Zhang, J. Struct. Chem., 2009, 50, 34–40.
- 38 Y. Sun and Y. Xia, J. Am. Chem. Soc., 2004, 126, 3892-3901.
- 39 Y. Sun and Y. Wang, Nano Lett., 2011, 11, 4386-4392.
- 40 M. R. Jones, K. D. Osberg, R. J. Macfarlane, M. R. Langille and C. A. Mirkin, *Chem. Rev.*, 2011, **111**, 3736–3827.
- 41 S. E. Skrabalak, J. Y. Chen, Y. G. Sun, X. M. Lu, L. Au, C. M. Cobley and Y. N. Xia, *Acc. Chem. Res.*, 2008, 41, 1587–1595.
- 42 X. Lu, H.-Y. Tuan, J. Chen, Z.-Y. Li, B. A. Korgel and Y. Xia, *J. Am. Chem. Soc.*, 2007, **129**, 1733–1742.
- 43 Y. Yin, C. Erdonmez, S. Aloni and A. P. Alivisatos, *J. Am. Chem. Soc.*, 2006, **128**, 12671–12673.
- 44 Karvianto and G. M. Chow, J. Nanopart. Res., 2012, 14, 1186.
- 45 J. Chen, J. M. McLellan, A. Siekkinen, Y. Xiong, Z.-Y. Li and Y. Xia, *J. Am. Chem. Soc.*, 2006, **128**, 14776–14777.