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# Extreme Resilience in Cochleate Nanoparticles

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# Extreme resilience in cochleate nanoparticles

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KEYWORDS: cochleate; membrane roll; force spectroscopy; nanomechanics; atomic force microscopy; myelin sheath

ABSTRACT Cochleates, prospective nanoscale drug delivery vehicles, are rolls of negatively-charged phospholipid membrane layers. The membrane layers are held together by calcium ions; however, neither the magnitude of membrane-interaction forces, nor the overall mechanical properties of cochleates have been known. Here we manipulated individual nanoparticles with atomic force microscopy to characterize their nanomechanical behavior. Their stiffness (4.2-12.5 N/m) and membrane-rupture forces (45.3-278 nN) are orders magnitude greater than those of the tough viral nanoshells. Even though the fundamental building material of cochleates is a fluid membrane, the combination of supramolecular geometry, the cross-linking action of calcium and the tight packing of the ions apparently lead to extreme mechanical resilience. The supramolecular design of cochleates may provide efficient protection for encapsulated materials

and give clues to understanding biomolecular structures of similar design, such as the myelinated axon.

## **Introduction**

Cochleates are membrane rolls made of dipalmitoyl phosphatidylserine and calcium ions, and are so named because of their cross-sectional geometry.<sup>1</sup> In a cochleate particle the spirally curving and stacked lipid layers are stabilized by calcium ions that cross-link the negatively charged lipid headgroups.<sup>1,2,3</sup> Cochleates have been proposed to be used as biocompatible pharmaceutical vehicles for molecules which are difficult to deliver such as amphotericin B to treat systemic mycoses,<sup>2,3,4,5,6</sup> factor VIII for replacement in haemophilia A,<sup>7,8</sup> or antigens for immunization.<sup>9,10,11,12,13,14,15</sup> In certain applications the active agent itself, rather than calcium, serves as the electrostatic glue that holds the membrane sheets together.<sup>16,17,18,19,20,21</sup> The structure of cochleates has previously been analyzed by using freeze fracture,<sup>1,3,12,22</sup> negative staining,<sup>7,23,24</sup> scanning electron<sup>12,21,25</sup> and light microscopies.<sup>3,10,12,21</sup> Although a detailed model has emerged about the geometry of cochleates, very little is known about their internal dynamics, mechanical properties and the structure-stabilizing forces. Considering that the cochleate displays a highly ordered structure in spite of being composed of a fluid membrane, its elasticity and viscosity need to be investigated experimentally. In the present work we mechanically manipulated individual cochleate nanoparticles by using atomic force microscopy and *in situ* force spectroscopy and found that they display unusually high stiffness and mechanical rupture forces.

## **Experimental**

### **Materials**

Dioleoyl phosphatidylserine (1,2-Dioleoyl-sn-Glycero-3-[Phospho-L-Serine], DOPS) was obtained from Avanti Polar Lipids, Inc. (Alabaster, Alabama, USA). Whatman Nuclepore membrane filters (d=100 nm, track-etched membranes), chloroform, sodium azide and 1 M calcium chloride solution were purchased from Sigma Aldrich Kft. (Budapest, Hungary). Argon 5.0 and nitrogen 5.0 gases were from Linde Gáz Magyarország Zrt. (Budapest, Hungary). Purified water was produced by a Milli-Q Integral 3 Water Production Unit (Merck Millipore, Billerica, MA, USA). Round mica sheets were obtained from Ted Pella, Inc. (Redding, CA, USA).

## **Methods**

### **Preparation of cochleates**

Cochleates were obtained from small unilamellar vesicles (SUV-s) prepared by the extrusion method.<sup>26</sup> Briefly, 500  $\mu$ l 20 mg/ml DOPS/chloroform stock solution was pipetted to a clean glass tube, and the organic solvent was evaporated under Ar gas stream. The lipid film was placed in vacuum (<20 Hgmm) for 30 min. to remove organic solvent traces, then hydrated with 2 mM NaN<sub>3</sub> in 100  $\mu$ l aliquots to 1 ml final volume during vigorous vortexing. The resultant milky suspension (lipid concentration: 10 mg/ml, 12.4 mM DOPS) was extruded 41 times through polycarbonate membrane filter with 100 nm pore diameter in an AvantiPolar Mini Extruder (Avanti Polar Lipids, Inc., Alabaster, AL, USA) at 30 $\pm$ 0.2°C temperature. The formation of cochleates was induced by adding a calcium stock solution (12.4 mM CaCl<sub>2</sub>, 2mM NaN<sub>3</sub>) dropwise to the SUV preparation until an equimolar ratio of Ca<sup>2+</sup> and DOPS was reached. The final lipid concentration was ~6.2 mM (5 mg/ml). The SUV sample turned opaque upon

Ca<sup>2+</sup> addition, and visible precipitation occurred. The sample was then incubated at 25±1°C for 1 week.

### **AFM imaging and force spectroscopy**

50 µl precipitated pellet was harvested from the cochleate sample and diluted with 200 µl purified water. 50 µl of this sample was applied to the surface of freshly cleaved mica and, after 1 minute incubation, washed gently with purified water and dried in a stream of N<sub>2</sub> gas of high purity. Cochleates were imaged with a Cypher instrument (Asylum Research, Santa Barbara, CA) with 0.4 to 1 Hz line-scanning rate in air. A silicon cantilever (OMCL AC-160TS, Olympus, Japan) was used in non-contact mode, oscillated at its resonance frequency (300-320 kHz, typically). Force spectra were taken in contact mode at defined locations of previously scanned areas and at constant 1 µm/s velocity. The cantilevers (stiffness typically around 35 N/m) were calibrated by the thermal method.<sup>27</sup> Temperature during the measurements was 29±1°C.

### **Image analysis**

Images were analyzed by using the built-in algorithms of the AFM driving software (IgorPro, WaveMetrics Inc., Lake Oswego, OR). AFM amplitude contrast images are shown in this paper. To determine height variations, height contrast data were used.

### **Statistical analysis of data**

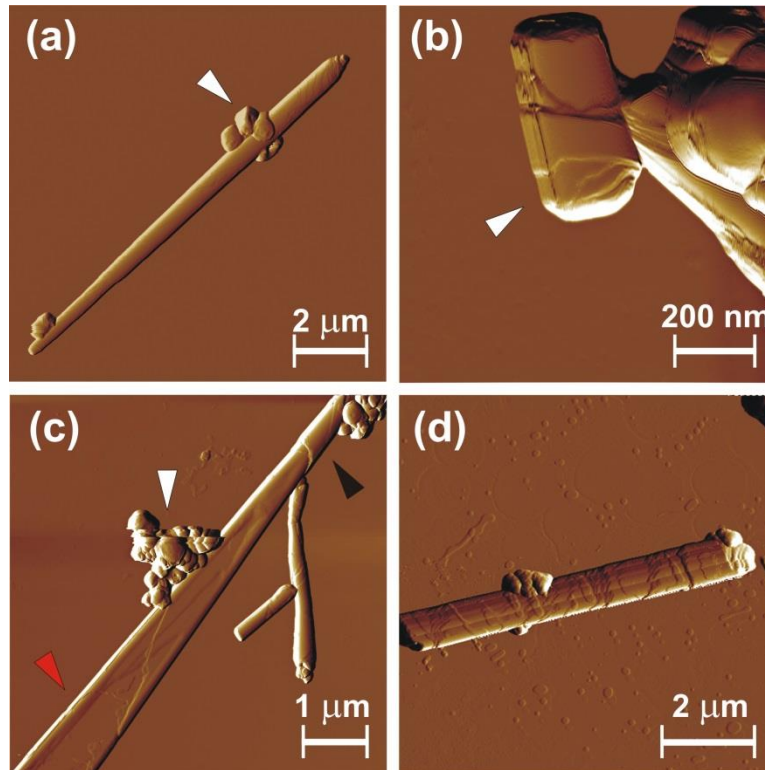
For phase “b” and “d” peak number analysis (**Figure 4. b**) only four-phase curves (n= 18) were used. The selected force spectra were used for peak-to-peak analysis (**Figure 5. a**). Step size data (**Figure 5. e**) were gathered from 8 individual cochleates. For stiffness determination (**Figure 6.**

**a)** not only complete but partial force curves (i.e., ones containing a complete phase “a” and partial “b”) were used as well (total n=40). To assemble the force-peak histogram (**Figure 6. b**), the same 18 spectra were analyzed that were used for peak number analysis. Scatter plots and histograms were created in Origin 7 software (OriginLab Ltd., Northampton, Ma, USA).

## **Results and Discussion**

### **Morphology of cochleates**

Atomic force microscopy of surface-adsorbed cochleate nanoparticles revealed a roll-like topographical appearance with varying axial and cross-sectional dimensions (**Figure 1. a and b**). Their overall structural features appeared consistent with the model suggested previously based on freeze-fracture electron microscopy observations.<sup>1</sup> The diameter of the rolls usually varied gradually along the longitudinal axis because the edge of the underlying lipid sheets extended to different distances (see **Figures 1, 2. c and 5.b**). Liposome aggregates often associated with the cochleates (see **Figures 1. a, c, 2. c and 5. b**). Certain rolls flattened on the surface (**Figure 1. c**) at one end with the other end remaining cylindrical. Such a capacity for flattening points at the presence of cavity inside the roll. While cochleates are usually thought of as compact structures with no internal aqueous space,<sup>28</sup> our observations indicate that under some circumstances a hollow core may be present. This observation supports recent electron microscopy findings.<sup>29</sup> Occasionally, ribbed cochleates were observed (**Figure 1. d**) which resembles multilamellar protein-bearing cochleates seen earlier with freeze fracture electron microscopy.<sup>12</sup> The topographical appearance suggests that axial compressive forces may have led to the formation of riblike surface ridges.



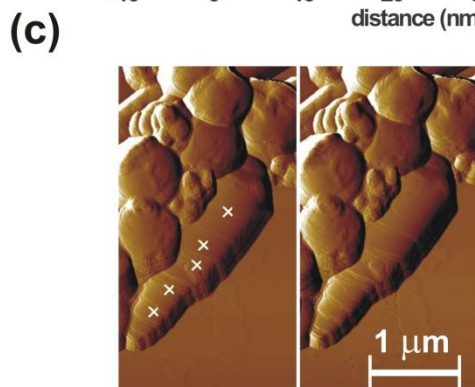
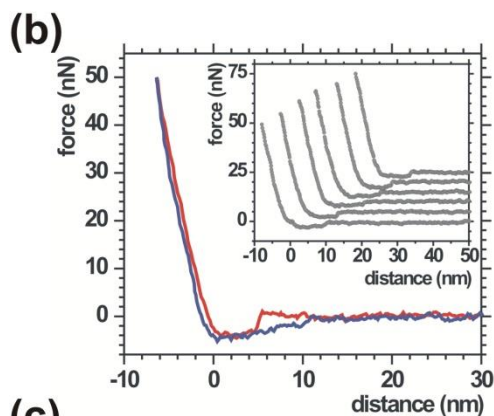
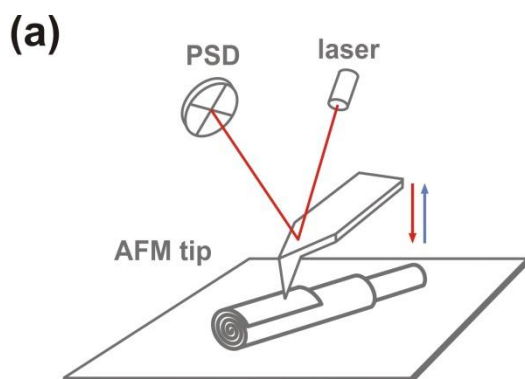
**Figure 1.** Morphology of cochleates. (a) Needle-like lipid roll associated with a few liposomal particles attached (white arrowhead). (b) Cochleate particle (white arrowhead) displaying topographical steps corresponding to superimposed and staggered lipid layers. (c) Cochleate flattened at one end (red arrowhead) and compacted at the other (black arrowhead). White arrowhead points at liposomal aggregate. (d) Cochleate with surface ridges.

## Force spectroscopy

### Reversible deformation of cochleates

We mechanically manipulated individual cochleate nanoparticles by pressing the AFM tip into their surface at specific locations then pulling the cantilever back at a constant velocity (**Figure 2. a**). When cochleates were loaded with forces up to 50 nN, no hysteresis was seen between the approach and retraction data (for typical force curves, see **Figure 2. b**) indicating that the

deformation was reversible on the time scale of a single mechanical cycle ( $\sim 10$  ms). To check for reversibility on a longer (minute) time scale, we loaded cochleates in 100 successive mechanical cycles at the same location. Neither changes in the slope of the force curves, nor the appearance of transitions were apparent (**Figure 2. b** inset). Furthermore, no permanent depressions on the cochleate surface were observable as a result of the manipulation (**Figure 2. c**). Thus, cochleates are resilient structures able to bear successively applied high forces without material fatigue or irreversible deformation.

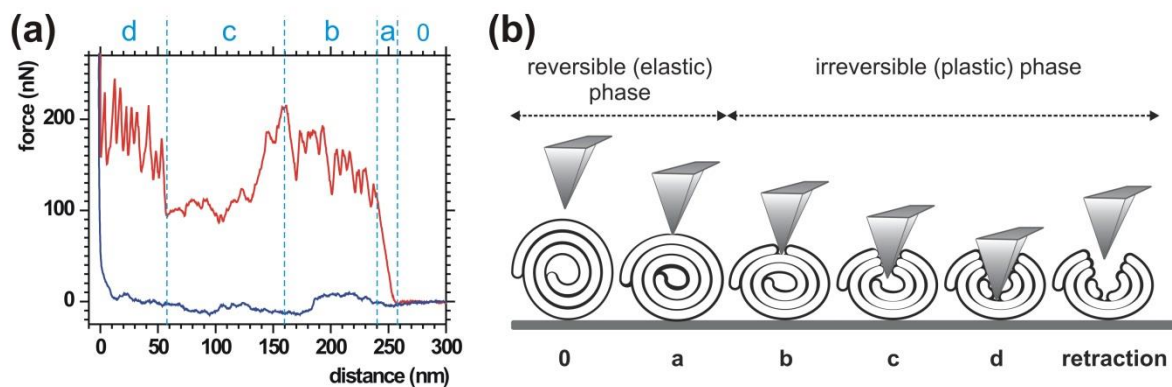




**Figure 2.** (a) Schematics of the force spectroscopy experiment. Force was obtained from the bending of a calibrated cantilever as a function of the distance traveled towards the substrate surface (PSD: position sensitive diode). (b) Approach (red) and retraction (blue) force curves collected at a maximum 50 nN loading force. Inset shows the first, 20<sup>th</sup>, 40<sup>th</sup>, 60<sup>th</sup>, 80<sup>th</sup> and 100<sup>th</sup> approach curves (gradually shifted for better display) of a 100-curve mechanomanipulation series taken from the same location. (c) Cochleate prior to (left image) and following (right image) force spectroscopy. 100 cycles at max. 50 nN force were carried out at each point marked with white crosses.

### Unique nanomechanical fingerprint revealed by pressing with excessive forces

Upon increasing the maximum load above a critical value, irreversible transitions became apparent in the force spectra, and a large hysteresis occurred between the approach and retraction traces. A typical force curve is shown in **Figure 3. a**.



**Figure 3.** (a) Representative force spectrum. Red and blue lines correspond to data collected during cantilever approach and retraction, respectively. Distance in nm denotes the position of

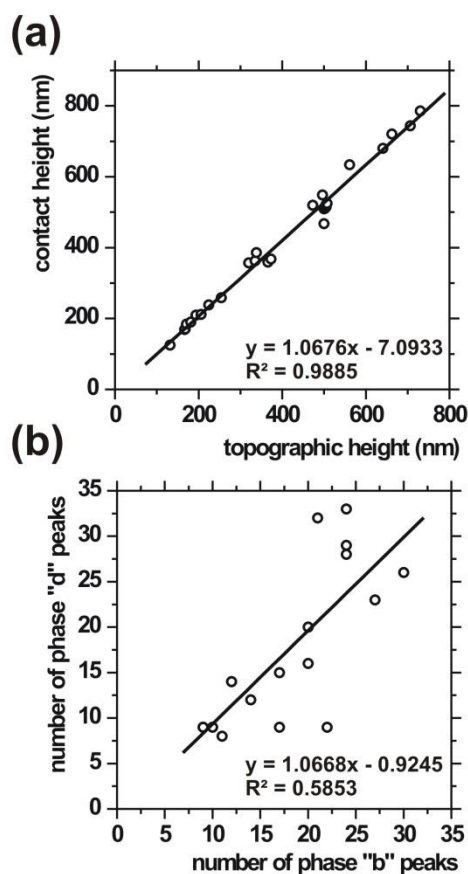
the cantilever tip relative to the mica surface. (b) Model of cochleate mechanomanipulation. Letters below the diagram correspond to phases shown in (a).

The approach curve could be divided into four distinct phases. After a featureless, constant-force region that corresponds to the unloaded movement of the cantilever towards the cochleate surface (indicated by “0”) a steep force rise emerged (phase “a”) that turned into a series of sawtooth-like peaks (phase “b”). This period ended with a transient force drop and an unstructured region (phase “c”) that transformed into another series of sawtooth force peaks (phase “d”) before force abruptly increased. Retraction curves were featureless on the force-scale of the approach, suggesting that in this phase there was no significant mechanical load on the cantilever. The well-structured and highly reproducible shape of the force spectra reflects a complex fingerprint that carries information about both the construction and the nanomechanical properties of the cochleate. In region “0” there is no contact between the tip and the sample, therefore the force is zero (see graphical interpretation of phases in **Figure 3. b**). Upon contact, the tip applies pressure on the membrane roll, therefore the cantilever bends and force rises steeply (phase “a”). Linearity of this region implies that the cochleate displays Hookean elasticity. The ideally elastic behavior is also substantiated by reversibility in this regime (see **Figure 2.**). Successive peaks of phase “b” likely correspond to the rupture of individual phospholipid bilayers. Once the tip reaches the hollow core of the membrane roll, resistance drops, hence force decreases and the curve becomes devoid of distinct features (phase “c”). Upon pushing the cantilever tip further, force begins to increase and successive peaks re-appear (phase “d”), implying that the membrane layers in the bottom half of the cochleate are reached and are broken through. Finally, the abrupt force increase at the end of phase "d" is likely caused by the cantilever tip reaching the mica surface, suggesting that the cochleate was, in this experiment,

pierced through its entire diameter. In some cases, phase “c” was absent and phases “a” and “d” merged, which may be explained by the lack of hollow core.

To validate the molecular events underlying the consensus force spectrum, we carried out detailed analyses of its features and compared them with topographical data.

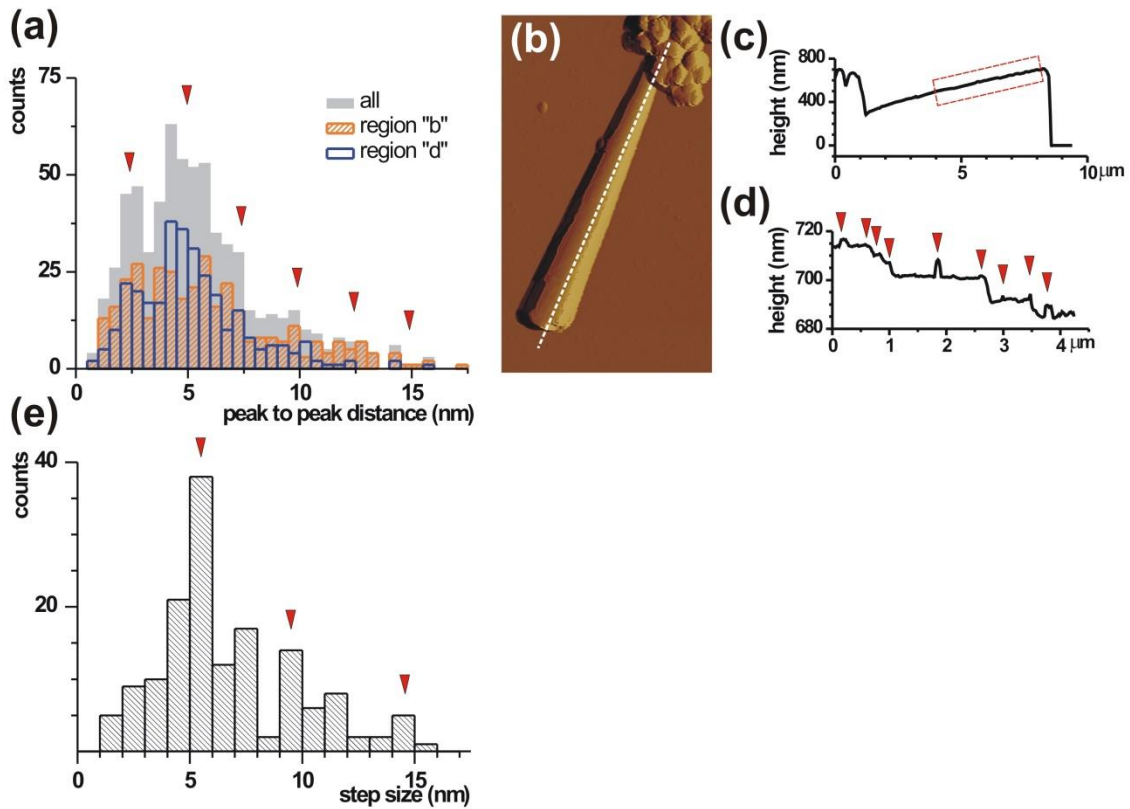
To test whether during nanomanipulation the cochleate was pierced through its full diameter, we compared the contact height (sum of phases "a" through "d" in **Figure 3.**) with the topographical height at the site of the manipulation obtained from AFM images. We found a very close correlation between these parameters (**Figure 4. a**), indicating that the force spectrum indeed reflects the events associated with a cantilever tip breaking through the entire stack of membrane layers within the cochleate nanoparticle. This conclusion furthermore predicts that phases "b" and "d" display similar number of force peaks, because the top and bottom halves of the cochleate are expected to contain similar numbers of membrane layers. In support, we found a good correlation between the numbers of phase "b" and "d" force peaks (**Figure 4. b**). Deviation from a closer correlation may be caused by missing force peaks due to sudden, simultaneous rupture of more than one superpositioned membrane layer, in any half of the cochleate.



**Figure 4.** Correlation between (a) contact and topographic heights of cochleates (n=29) and (b) the number of peaks in phases „b” and „d” of the force curves (n=18).

To test whether phase “b” and “d” force peaks correspond to membrane layer breakthrough transitions, we measured the distribution of distance gain between consecutive peaks (**Figure 5. a**). The peak-to-peak distance histograms of both “b” and “d” phases display an apparently multimodal distribution (**Figure 5. a**). The highest peak is at approximately 5 nm which corresponds well to the thickness of a single phospholipid bilayer.<sup>30, 31</sup> Accordingly, most of the transitions manifested in a force sawtooth correspond to events during which a single bilayer was mechanically broken through. Interestingly, further maxima appeared at integer multiples of ~2.5

nm, suggesting that monolayers or layer combinations can also be broken through in a single transition. This finding supports our interpretation of the deviation from perfect correlation in the peak number analysis (**Figure 4. b**). If the histogram peaks correspond to the distance gain caused by breaking through the lipid layer components of the cochleate, then a similar topographical step height distribution may be observed on the surface of the nanoparticle. We tested this hypothesis by a detailed surface topography analysis (**Figures 5. b-d**). The step height distribution showed a major peak between 5 and 6 (**Figure 5. e**), which is in excellent agreement with the distribution of peak-to-peak distances implying that it is indeed monolayers and bilayer combinations that are broken through during mechanomanipulation of cochleates. Altogether, the mechanical fingerprint (**Figure 3. a**) displays the elastic and plastic deformations of the cochleate nanoparticle. The plastic deformation is thus caused by the sequential rupture of the component lipid membrane sheets in integer multiples of a monolayer.



**Figure 5.** (a) Distribution of peak-to-peak distances in the different regions of the force curves ( $n_b=356$ ,  $n_d=313$ ). Red arrowheads point at histogram modes. (b) Amplitude-contrast image of a cochleate. The white dashed line shows the section along which a detailed topographical analysis was carried out (c-d). (c) Topographical height distribution along the cochleate nanoparticle in (b). Framed region is magnified further in (d). (d) Topographical height profile along the cochleate axis corrected for overall baseline slope. Red arrowheads point at height steps. (e) Histogram of topographical step sizes with modes marked with red arrowheads ( $n=166$ ).

### Stiffness and mechanical resistance

The magnitude of forces necessary for either the elastic or plastic deformation reflects the mechanical properties of the cochleate and is related to the stability of the nanoparticle. The slope of the force curve's linear region (phase "a") corresponds to the stiffness (spring constant) of the membrane structure according to Hooke's law of elasticity.<sup>32</sup> The spring constant of cochleates is surprisingly large ( $8.2 \pm 1.5$  N/m, see **Figure 6. a**), orders of magnitude greater than that of empty viral capsids, liposomes, supported lipid bilayers or microtubules (**Table 1**). Considering that the lipid bilayer displays fluidlike behavior, the observation of such an extreme stiffness is unexpected.

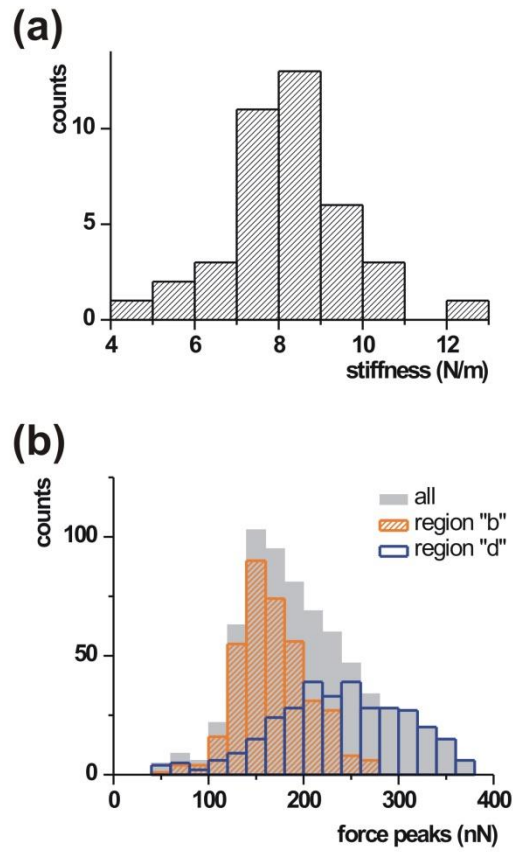
Table 1. Mechanical properties of ordered nanoscale biomolecular systems

	Spring constant (N/m)	Rupture force (nN)	References
Empty viral capsids	0.11 - 1.03	0.6 - 5.8	33, 34, 35, 36, 37
Liposomes	0.01 - 0.12	0.6 - 1.1	38, 39
Supported lipid bilayers	0.01 - 1	0.4 - 36	40, 41, 42
Microtubules	0.074 - 0.1	0.3 - 0.5	43, 44

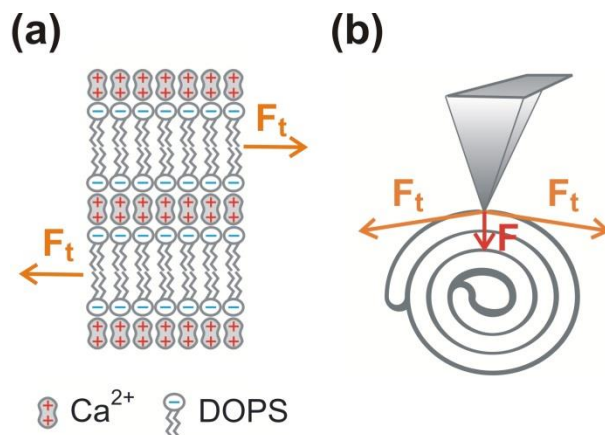
Furthermore, cochleate membrane layers could withstand forces up to a few hundred nN prior to rupture (**Figure 6. b**). By comparison, the force needed to rupture a covalent bond is a few nN ( $1.4 \pm 0.3$  nN for S-Au and  $2.0 \pm 0.3$  nN for Si-C bonds, respectively).<sup>45</sup> Highly ordered protein structures, such as microtubules or viral capsids suffer failure upon exposure to a few hundred pN to few nN forces. Even planar lipid bilayers supported by hard substrate break through at forces below a few tens of nN depending on the type of AFM tip, lipid composition, temperature and buffer conditions (see **Table 1**). Calcium may increase the mechanical resistance of phospholipid bilayers<sup>46</sup> and it has a pronounced effect on negatively charged lipids.<sup>40, 47</sup> However, the lipid headgroup ordering effect of calcium ions cannot solely account for the anomalously tough mechanical properties of cochleates. Most plausibly a combination of the

onion-shell geometry, the local cross-linking of lipid headgroups between neighboring membrane sheets by calcium ions and the parallel coupling of these interactions along the plane of the membrane leads to the extreme mechanical stability and stiffness. Because of the parallel coupling of calcium-based cross-links between neighboring lipid bilayers, a high shear force is required for the displacement of a membrane on top of an underlying one (**Figure 7. a**). In a nanomechanical experimental setting this shear force is the tangent-vector component of the deforming force (**Figure 7. b**), explaining why high forces are required for both elastic and plastic deformations. Furthermore, we hypothesize that the tight packing of calcium ions between the bilayers places a strong constraint on lateral diffusion so much that a quasi-crystalline phase is formed. During plastic deformation the membrane sheets are shifted by the high deforming forces, but because of the quasi-crystalline structure restoring forces do not arise. Thus, the deformation is expected to persist.





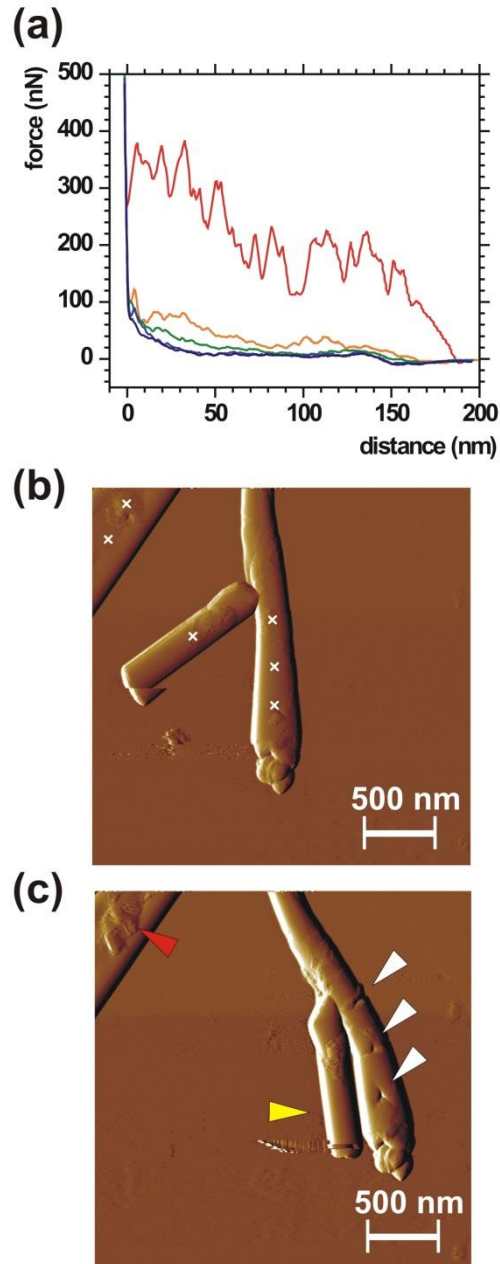
**Figure 6.** (a) Histogram of cochleate stiffness ( $n=40$ ). (b) Histogram of peak forces ( $n_b=373$  and  $n_d=330$ ).



**Figure 7.** (a) Schematics of the cross-sectional structure of cochleate membrane layers.  $F_t$  corresponds to the shear force necessary for membrane-sheet displacement. (b) Schematics of AFM mechanomanipulation illustrating the tangential components of the deforming force.

### **Plastic deformation of cochleates**

To test the prediction of irreversible plastic deformation, we exposed a cochleate nanoparticle to repeated cycles of loading with excessive forces (**Figure 8.**). We found that only the first mechanical cycle displayed the characteristic mechanical fingerprint, and the subsequent cycles were essentially devoid of transitions and hysteresis (**Figure 8. a**). Furthermore, the plastic deformation resulted in a permanent depression in the cochleate surface that could be detected by AFM imaging (**Figures 8. b-c**).



**Figure 8.** Plastic deformation of cochleates. (a) Successive approach curves from the first (red) to the fifth (violet) in temporal order. Cochleates prior to (b) and following (c) force spectroscopy at the points marked with white crosses. Structural rearrangements due to mechanomanipulation are marked with colored arrowheads. Red arrowhead points at a site where membrane sheets bulged out of the cochleate surface. White arrowheads point at surface

depressions. Yellow arrowhead points at a cochleate which was repositioned on mica as a result of the nanomechanical manipulation.

## **Conclusions**

In summary, AFM imaging and force spectroscopy uncovered both the topological features and surface-hidden structural details of phospholipid membrane rolls. Cochleates not only withstand forces up to 100 nN without rupture, but within this force range they display ideal elasticity. Upon exposure to forces well exceeding 100 nN, a characteristic nanomechanical fingerprint emerges with transitions corresponding to the sequential breaking through lipid layers. Because of calcium-dependent coupling within the onion-shell-like cross-sectional structure of the cochleate, a quasi-crystalline order is maintained the mechanical distortion of which results in plastic deformation upon exposure to extreme forces. The results presented in this work were obtained on partially hydrated cochleates under ambient humidity conditions, and their properties in fully submerged aqueous conditions are yet to be characterized. However, cochleates have emerged here as resilient nanoparticles with extreme mechanical properties far exceeding those of tough viral nanoshells and complex molecular architectures designed for mechanical resistance. Thus, the geometry and the internal interactions of cochleates may make them ideal for mechanical protection. Although in myelinated sheaths protein molecules rather than calcium ions stabilize the underlying membrane sheets, because of their structural similarity to cochleates the myelinated sheath may, intriguingly, also function as a mechanical barrier for the enveloped neuronal axon. In a technological setting cochleates, as pharmaceutical vehicles, may provide unparalleled protection for the molecular species to be carried harmlessly towards its destination.

**Supporting Information.** Supporting Information Available: Cochleate internal cavity analysis; Force curves. This material is available free of charge via the Internet at <http://pubs.acs.org>.

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### **Author Contributions**

The manuscript was written through contributions of all authors. All authors have given approval to the final version of the manuscript.

### **Notes**

The authors declare no competing financial interest.

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## ABBREVIATIONS

AFM, atomic force microscope; DOPS, dioleoyl phosphatidylserine; PSD, position sensitive diode; SUV, small, unilamellar vesicle.

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