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# POSITRONIUM DEUTERIDE AND HYDRIDE IN MgO CRYSTALS

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Low-temperature positron lifetime and Doppler broadening measurements were made in MgO crystals containing  $D^-$  or  $H^-$  ions in order to investigate the temperature dependence of the positron trapping by  $D^-$  and  $H^-$  ions and elucidate the possible formation of PsD (PsH) states. Positrons are trapped at  $D^-$  and  $H^-$  ions once the oxygen vacancies, which are more effective positron traps, are eliminated by annealing the crystals at high temperatures in a reducing atmosphere. From the temperature dependence of the annihilation parameters the positron trapping coefficients for  $D^-$  and  $H^-$  centers were shown to increase with temperature between 100–300 K. The lifetime of the PsD (PsH) state into D (H) and Ps is also temperature independent. No isotopic effect was observed.

Theoretical and experimental investigations of the formation of simple positronium compounds are necessary in order to use both positron and positronium (Ps) as chemical probes. In the past positron-anion bound sates  $[e^+X^-)$  or positronium compounds of the type PsX, have been theoretically investigated.<sup>1,2</sup> Calculations demonstrate that certain light PsX molecules are stable and their annihilation characteristics can be predicted. However, the detection of these Ps-compounds is difficult and uncertain.

Although positron lifetime spectra from liquid solutions show lifetime components that may, in principle, be attributed to the formation of  $e^+$ - or Ps-compounds, it is very difficult to identify unambiguously which  $e^+$ - or Ps-complexes are responsible for these lifetime components. In addition, it is difficult to resolve the complex spectra into three or more components. Thus the analyses can provide only limited quantitative information.

PsH has been observed when a thermalized positron is trapped by an H<sup>-</sup> ion (a proton with two electrons substituting for an indigenous  $O^{2-}$  ion) in MgO crystals,<sup>3</sup> and as a result of collisions between positrons and CH<sub>4</sub> molecules.<sup>4</sup> Thermochemically reduced hydrogen doped MgO crystals are ideal to investigate PsH because (1) the positron annihilation characteristics in these crystals are known, (2) the positron lifetime spectrum is simple, and (3) the concentration of H<sup>-</sup> ions and other lattice defects that

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trap positrons can be monitored by optical absorption measurements. Ps is observed in MgO containing  $H^-$  ions only after stronger positron traps such as neutral anion vacancies (F centers) are annealed out.

In a previous paper,<sup>3</sup> we have determined the lifetime of the PsH in MgO, and its dissociation rate into Ps and H. In this work, the temperature effect on the formation of both positronium hydride (PsH) and deuteride (PsD), and on their annihilation parameters is studied. In addition, the possibility of an isotopic effect on the annihilation characteristics is investigated.

### Experimental

The MgO crystals used were grown by the arc fusion method at he Oak Ridge National Laboratory using high purity-grade MgO powder from the Kanto Chemical Company, Tokyo (Japan). Large concentrations of hydrogen and deuterium in the crystals were obtained by presoaking MgO powder in H<sub>2</sub>O and D<sub>2</sub>O, respectively. In order to produce H<sup>-</sup> and D<sup>-</sup> ions, the crystals were subjected to severe thermochemical reduction (TCR) at high temperature (~2400 K) and under high pressure Mg or Ca vapor (~7 atm) in a tantalum chamber, followed by rapid cooling.<sup>5,6</sup> This TCR treatment produces a high concentration of F centers. A proton (deuteron) can be trapped at one of these vacancies resulting in an H<sup>-</sup> (D<sup>-</sup>) ion. F centers produce an optical absorption band at 250 nm, while H<sup>-</sup> ions can be monitored by local vibrational modes at 1053, 1032 and 1024 cm<sup>-1</sup>.<sup>7</sup> The absorption bands for D<sup>-</sup> lie in the reststrahlen and can not be observed. The concentration of F centers and H<sup>-</sup> (D<sup>-</sup>) ions are determined from optical absorption measurements.

Deuterium doped crystals always contain hydrogen. The ratio between the H<sup>-</sup> ion and the D<sup>-</sup> ion concentrations was estimated to be 1 : 4. In as-grown crystals, the same ratio is observed between the intensity of the 2445 cm<sup>-1</sup> and the 3296 cm<sup>-1</sup> bands associated with  $V_{\rm OD}$  and  $V_{\rm OH}$  centers, respectively. A  $V_{\rm OD}$  center has the following linear configuration OD<sup>-</sup>—[Mg vacancy]—O<sup>2-</sup>. Replacing deuterium for hdrogen gives the  $V_{\rm OH}$ .

Positron lifetime measurements were performed in the temperature range 100–300 K, with the samples inside a liquid-N<sub>2</sub> flow cryostat. The spectra were obtained with a 235 ps resolution (FWHM) spectrometer. The lifetime spectra were analyzed by means of the POSITRONFIT program. Doppler-broadening experiments were performed over a temperature range of 10–300 K in an optical He-closed cycle cryostat. The annihilation radiation was recorded with a Ge detector with a resolution of 1.62 keV at 1.33 MeV line of <sup>60</sup>Co. The Doppler broadening of the annihilation peak was characterized by the lineshape parameter S describing the fraction of annihilation events in an energy window of (511.0 ± 0.8) keV. A <sup>22</sup>Na positron source sandwiched between a pair of samples was used.

# **Results and discussion**

### Doppler broadening measurements

The S parameter for as-grown and thermochemically reduced samples, intentionally doped with H or D, shows a weak temperature dependence. However, as the F center content in the thermochemically reduced samples is lowered by successive reducing



Fig. 1. Difference  $[S(T)-S_{10K}]$  versus temperature for a pair of MgO: D<sup>-</sup> samples;  $\blacksquare$  As-TCR with  $[F] = 1.0 \cdot 10^{-6}$  at<sup>-1</sup>,  $\blacklozenge$  after annealing at 1380 K for 30 min,  $[F] = 1.4 \cdot 10^{-8}$  at<sup>-1</sup>,  $\blacklozenge$  after annealing at 1440 K for 10 min, without F centers, and O after annealing at 1870 K for 45 min in an oxydizing atmosphere, without F, D<sup>-</sup> and H<sup>-</sup> centers. The D<sup>-</sup> and H<sup>-</sup> concentrations for  $\blacksquare$ ,  $\blacklozenge$  and  $\blacklozenge$  were  $7.4 \cdot 10^{-6}$  and  $1.9 \cdot 10^{-6}$  at<sup>-1</sup>, respectively

treatments at 1380 K in a graphite container, this dependence slowly increases. Figure 1 shows the difference  $[S(T) - S_{10K}]$  versus temperature for a pair of thermochemically reduced D-doped MgO crystals;  $S_{10K}$  is the S value measured at 10 K. After annealing out all H<sup>-</sup> centers in flowing oxygen at 1870 K, the temperature dependence is very weak, and  $[S(T) - S_{10K}]$  shows the same temperature dependence as a pair of as-grown crystals. The same behavior is observed for a pair of thermochemically reduced H-doped samples.

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### Positron lifetime measurements

After TCR the positron lifetime spectra of MgO crystals doped with H or D exhibit two components, but they become three-component spectra when the F center concentration is lowered by reducing treatments. Figure 2 depicts the mean positron lifetime  $\langle \tau \rangle$  as a function of temperature for a pair of D-doped samples. The mean lifetime of the thermochemically reduced H-doped samples shows the same behavior.



Fig. 2. Mean positron lifetime versus temperature for a pair of MgO: D<sup>-</sup> samples;  $\blacksquare$  As-TCR with [F] = 1.0  $\cdot$  10<sup>-6</sup> at<sup>-1</sup>,  $\blacklozenge$  after annealing at 1380 K for 30 min, [F] = 1.4  $\cdot$  10<sup>-8</sup> at<sup>-1</sup>, and  $\blacklozenge$  after annealing at 1440 K for 10 min, without F centers. The D<sup>-</sup> and H<sup>-</sup> concentrations were 7.4  $\cdot$  10<sup>-6</sup> at -1, respectively

For thermochemically reduced samples, the lifetime values of the second component,  $\tau_2$ , are essentially temperature independent:  $(250 \pm 50)$  ps and  $(350 \pm 30)$  ps in the Hand D-doped samples, respectively. For both types of samples the increase of  $\langle \tau \rangle$  with temperature is due to an increase in the intensity of the second component  $I_2$ . This second component could be attributed to positron trapping at F centers because the  $\tau_2$ value gradually increases up to  $\sim 520$  ps when the concentration of F centers is lowered by reducing treatments at 1380 K. However, we have found that the  $\tau_2$  value for thermochemically reduced crystals changes with the H or D content and the F center concentration. The higher the H or D content and F concentration, the higher the  $\tau_2$ value. This suggests that the positron traps responsible for this second component are not only F centers but also some complex defects formed by association of anion vacancies and protons or deuterons.

The reducing treatments produce dramatic changes in the lifetime spectra. When the F center concentration decreases to about  $10^{15}$  cm<sup>-3</sup> and the H<sup>-</sup> or D<sup>-</sup> concentration is  $10^{18}$  cm<sup>-3</sup>, the spectra becomes three-component. The lifetime values of the second and third components,  $\tau_2$  and  $\tau_3$  are temperature independent although their values gradually



Fig. 3. Intensities  $I_2$  and  $I_3$  versus temperature for a pair of MgO: D<sup>-</sup> samples, without F centers, containing D<sup>-</sup> and H<sup>-</sup> concentrations of  $7.4 \cdot 10^{-6}$  and  $1.9 \cdot 10^{-6}$  at<sup>-1</sup>, respectively

increase as the F center concentration decreases. After annealing out all the F centers, the  $\tau_2$  value is  $(510 \pm 30)$  ps and  $(500 \pm 30)$  ps for the H- and D-doped samples, respectively. The lifetime  $\tau_3$  value is  $(2.2 \pm 0.2)$  ns for both types of samples. In all cases, the intensities  $I_2$  and  $I_3$  increase with temperature as shown in Fig. 3.

The above results indicate that the temperature dependence of the annihilation parametes S and  $\langle \tau \rangle$  is due to an increase of the positron trapping at the defects responsible for the second and third components. In thermochemically reduced H-doped MgO crystals, it has been demonstrated that H<sup>-</sup> ions are positron traps forming PsH states. The second component is attributed to the annihilation of these states, and the third component to annihilation of ortho-Ps states created by the PsH dissociation into Ps and H. Since no differences between the results for the H- and D-doped samples are found, it can be concluded that D<sup>-</sup> ions trap thermalized positrons forming PsD states.





Using the equation of the positron decay model for MgO containing H<sup>-</sup> ions and no F centers<sup>3</sup> (see Appendix), we have determined (1) the dissociation rates for the PsH (PsD) states, (2) their lifetime  $\tau_{PsH}(\tau_{PsD})$  and (3) the trapping rates,  $\kappa_{H^-}(\kappa_{D^-})$ , at H<sup>-</sup> and D<sup>-</sup> versus temperature. For the temperature range 100–300 K, the dissociation rates and the lifetime for PsH or PsD are  $(1.42 \pm 0.12) \cdot 10^8 \text{ s}^{-1}$  and (650 ± 30) ps, respectively. Nevertheless, the positron trapping coefficient for H<sup>-</sup> or D<sup>-</sup> increases with temperature as shown in Fig. 4. Assuming  $\kappa_{H^-} = \mu_{H^-}C_{H^-}$  (or  $\kappa_{D^-} = \mu_{D^-}C_{D^-}$ ) the positron trapping coefficient for H<sup>-</sup> or D<sup>-</sup> ions can be calculated using the H<sup>-</sup> (D<sup>-</sup>) concentration  $C_{H^-}(C_{D^-})$  obtained from optical absorption measurements. The trapping coefficients are also shown in Fig. 4. Although the PsH lifetime is about 12% higher than the value previously reported,<sup>3</sup> the dissociation rate and trapping coefficient agree with those given in Reference 3.

The lifetime value of  $(650 \pm 30)$  ps is about 1.5 times higher than the more recent values predicted for PsH in vacuum.<sup>7,8</sup> We believe that the lifetime values predicted for PsH in vacuum cannot be extrapolated to PsH in ionic crystals because in the latter the PsH molecule must be polarized by the lattice; the polarization of the PsH molecule enhances its lifetime.

Finally, it should be noted that the slight increase of the positron trapping coefficient,  $\mu_{\text{H-}}(\mu_{\text{D-}})$ , with temperature is in agreement with the temperature dependence theoretically predicted for positively-charged vacancies in wide-gap crystals,<sup>9</sup> H<sup>-</sup> (D<sup>-</sup>) ions in MgO are positively charged centers relative to the lattice.

## Conclusion

We conclude that in MgO, PsD states are formed by trapping of thermalized positrons at D<sup>-</sup> ions. The lifetime of the PsH or PsD molecule, and their dissociation rate appear to be temperature independent and indistinguishable. The temperature dependence of both the S parameter and the mean positron lifetime is consistent with an increase of the positron trapping coefficient for H<sup>-</sup> (D<sup>-</sup>) ions with temperature, that is, with an increase of the PsH (PsD) formation with temperature.

### References

- 1. V. I. GOLDANSKII, At. Energy Rev., 6 (1968) 3.
- 2. D. M. SCHRADER, Materials Sci. Forum, 175-178 (1995) 77.
- 3. R. PAREJA, R. M. DE LA CRUZ, M. A. PEDROSA, R. GONZÁLEZ, Phys. Rev., B41 (1990) 6220.
- 4. D. M. SCHRADER, F. M. JACOBSEN, N. P. FRANDSEN, U. MIKKELSEN, Phys. Rev. Lett., 69 (1992) 57.
- 5. R. GONZÁLEZ, Y. CHEN, M. MOSTOLLER, Phys. Rev., B24 (1981) 6862.
- 6. Y. CHEN, R. GONZÁLEZ, O. E. SCHOW, G. P. SUMMERS, Phys. Rev., B27 (1983) 1276.
- 7. D. C. CLARY, J. Phys., B9 (1976) 3115.
- 8. Y. K. HO, Phys. Rev., A34 (1986) 609.
- 9. M. J. PUSKA, C. CORBEL, R. M. NIEMINEN, Phys. Rev., B41 (1990) 9980.